

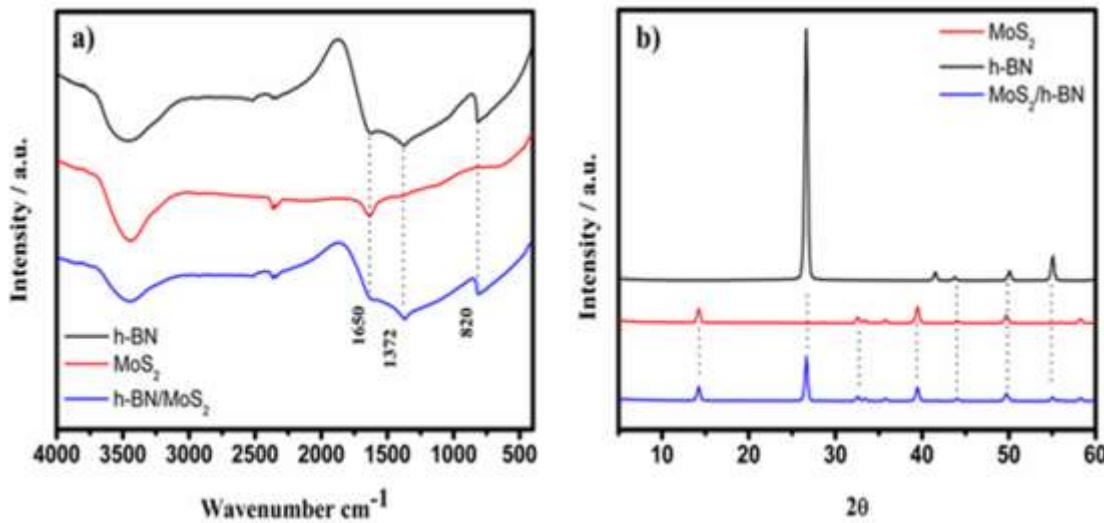
## SUPPORTING INFORMATION

# Hybrid MoS<sub>2</sub>/h-BN Nanofillers As Synergic Heat Dissipation and Reinforcement Additives in Epoxy Nanocomposites

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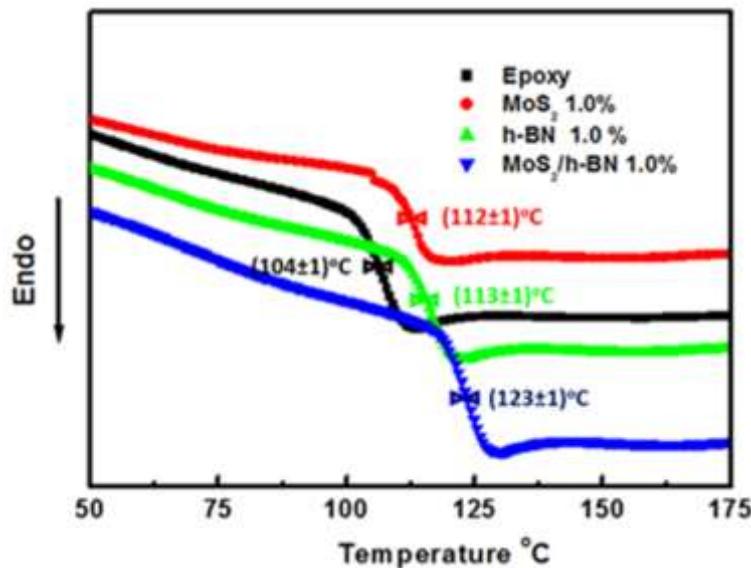


**Figure S1. (a) FTIR spectra and (b) XRD patterns of MoS<sub>2</sub>, h-BN and hybrid MoS<sub>2</sub>/h-BN material. The dash lines indicate the correspondence in the signals for the hybrid composite and the pure nanomaterials.**

The FTIR spectra of h-BN, MoS<sub>2</sub>, and mixture of h-BN/MoS<sub>2</sub> (Figure S1a), show the main characteristic peaks, which are considered fingerprints of sp<sup>2</sup> bonds of h-BN: at 1372 cm<sup>-1</sup> corresponding to the in-plane B–N transverse stretching and 820 cm<sup>-1</sup> corresponding to the out-of-plane B–N–B bending vibration.<sup>1</sup> The broad band around ~3428 cm<sup>-1</sup> can be attributed by the residual bonded N-H or caused to the water molecules absorbed on the surface of the samples.<sup>1-4</sup> The vibrational modes at ~3435 and 1650 cm<sup>-1</sup> are assigned to hydroxyl functionalities of adsorbed moisture on the MoS<sub>2</sub> nanosheets.<sup>5</sup>

The XRD patterns of h-BN, MoS<sub>2</sub>, and mixture of h-BN/MoS<sub>2</sub> are shown in Figure S1b. The intense diffraction peak  $2\theta \sim 26.76^\circ$ , (002) plane, denote the high crystallinity of the h-BN nanostructure.<sup>1</sup> MoS<sub>2</sub> showed characteristic peaks at  $2\theta$  of 12.00°, 32.00°, 40.00°, and 50.01° due to diffraction from (002), (100), (103), and (105) planes, respectively.<sup>5</sup>

The diffraction pattern was found to be fully matched with JCPDS file 37-1492. The sample of the heterostructure h-BN/MoS<sub>2</sub> showed peaks corresponding to both h-BN and MoS<sub>2</sub> separated samples.



**Figure S2. DSC curves of neat epoxy and composites containing 1.0 wt.% of MoS<sub>2</sub>, h-BN and hybrid MoS<sub>2</sub>/h-BN.**

DSC curves for pure epoxy and the composites containing 1.0 wt% of MoS<sub>2</sub>, h-BN and MoS<sub>2</sub>/h-BN in the second heating scan are shown in Figure S2. The glass transition temperature ( $T_g$ ) was determined from the midpoints of three measurements (average values) for each sample. The  $T_g$  values showed standard deviation of  $\pm 1^\circ\text{C}$ . Increases in  $T_g$  values for all systems were observed, however, the system containing 1.0 wt % of MoS<sub>2</sub>/h-BN showed the best increase of up to  $19^\circ\text{C}$  when compared to pure polymer. Generally, in epoxy systems the increase of the glass transition temperature is associated with restriction of the mobility of their polymer chains due to a high degree of cross-link and reduction of free volume.<sup>6</sup> Factors that affect the value of  $T_g$  for epoxy nanocomposites include: the degree of dispersion of the nanofillers, interaction between

nanofillers/polymer and curing process conditions. Under the conditions studied here, it can be said that the cross-linking process was not affected by the addition of nanofillers, i.e., the polymer chains were not generated with free ends as observed by other authors.<sup>7</sup> Moreover, it is possible that the interface between the introduced MoS<sub>2</sub> and h-BN served as chemical anchor points for epoxy chains, thus impacting the mobility thereof and increasing the  $T_g$  values in the hydride composite.

**Table S1. Storage modulus,  $T_g$  and cross-link density values of neat epoxy and nanocomposites with different wt% of 2D nanofiller**

System (wt%)	Storage Modulus at $T_g$ (MPa)	$T_g$ (°C) ±1	Cross-link Density (mol cm <sup>-3</sup> )
Neat epoxy	1662±13	123	0.5056±0,0011
MoS <sub>2</sub> 0.25	1642±11	125	0.5289±0,0015
MoS <sub>2</sub> 0.50	1725±21	126	0.5312±0,0008
MoS <sub>2</sub> 1.0	1738±19	126	0.5828±0.0014
h-BN 0.25	1613±22	127	0.5013±0.0013
h-BN 0.50	1645±27	128	0.5343±0.0010
h-BN 1.0	1637±29	130	0.6371±0.0011
MoS <sub>2</sub> / h-BN 0.25	2627±22	131	0.5917±0.0014
MoS <sub>2</sub> / h-BN 0.50	2674±26	133	0.6273±0.0012
MoS <sub>2</sub> / h-BN 1.0	2777±18	135	0.7183±0.0013

The values of brittleness ( $B$ ) for viscoelastic materials enables the comparative analysis of several types of polymer-based materials, metal, alloys, ceramics and glasses.<sup>8</sup>

The parameters obtained from tensile testing can be connected with the brittleness through the equation proposed by Brostow and Narkis.<sup>9</sup>

$$B = \frac{1}{(\varepsilon_b E')}$$
 (2)

where  $\varepsilon_b$  is the tensile elongation at break, and  $E'$  is the storage modulus determined by DMA test at 1 Hz, both at same temperature.<sup>8-10</sup>. The numerical values of storage modulus, tensile elongation and brittleness at 25°C for neat epoxy and nanocomposites are shown in Table S2.

**Table S2. Storage modulus ( $E'$ ), tensile elongation at break ( $\varepsilon_b$ ), and brittleness (B) values of neat epoxy and nanocomposites with different wt% of 2D nanofiller at 25°C**

System (wt%)	$E'$ (Pa)/ $10^8$	$\varepsilon_b$ (%)	B (%Pa/ $10^{10}$ )
Neat epoxy	44.49±09	15.28	0.147
MoS <sub>2</sub> 0.25	46.72±11	18.69	0.115
MoS <sub>2</sub> 0.50	46.35±13	20.27	0.106
MoS <sub>2</sub> 1.0	44.25±15	19.38	0.116
h-BN 0.25	48.28±07	19.83	0.104
h-BN 0.50	49.54±17	19.85	0.102
h-BN 1.0	50.21±14	17.56	0.113
MoS <sub>2</sub> / h-BN 0.25	56.68±10	18.33	0.093
MoS <sub>2</sub> / h-BN 0.50	55.24±15	21.76	0.081
MoS <sub>2</sub> / h-BN 1.0	59.97±12	23.25	0.074

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