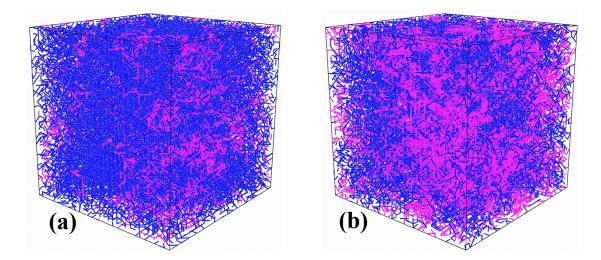
## **Supporting Information**

## Effect of Composition Asymmetry on the Phase Separation and Crystallization in Double Crystalline Binary Polymer Blends: A Dynamic Monte Carlo Simulation Study

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Figure S1. Snapshots at  $U_p = 0$  for various compositions,  $x_B$ : (a) 0.125, (b) 0.375, (c) 0.5, (d) 0.625 and (e) 0.875. Blue and magenta colors represent bonds of A- and B-polymers, respectively.



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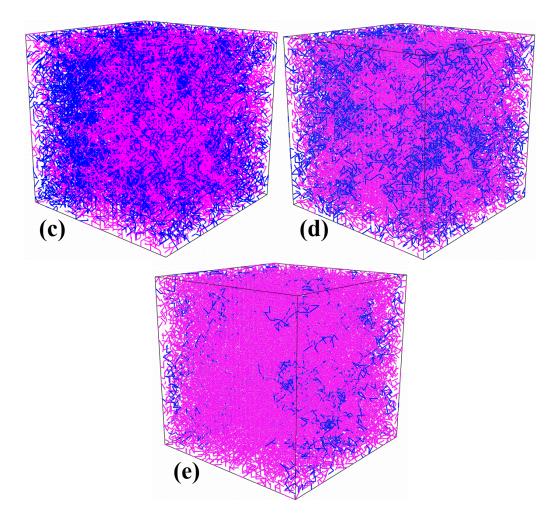


Figure S2. Change in mean square displacement of center of mass  $(d_{cm}^2)$  with  $U_p$  for Aand B-polymers: (a)  $d_{cm_A}^2$  vs.  $U_p$  at  $\lambda = 1$ , (b)  $d_{cm_A}^2$  vs.  $U_p$  at  $\lambda = 6$ , (c)  $d_{cm_B}^2$  vs.  $U_p$  at  $\lambda$ = 1, (d)  $d_{cm_B}^2$  vs.  $U_p$  at  $\lambda = 6$ .

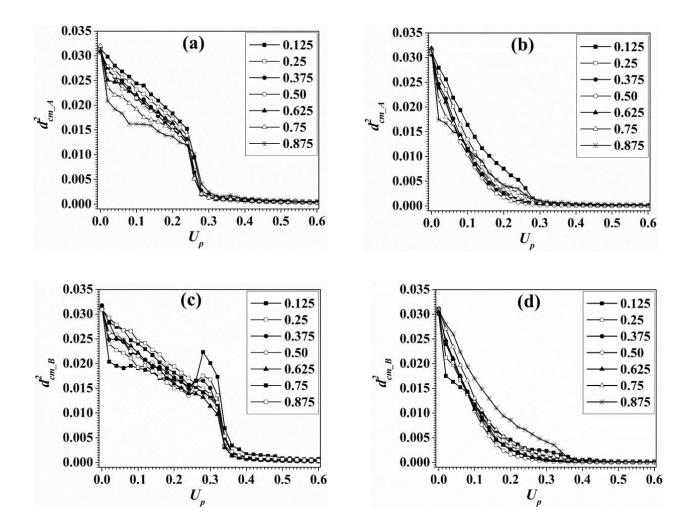
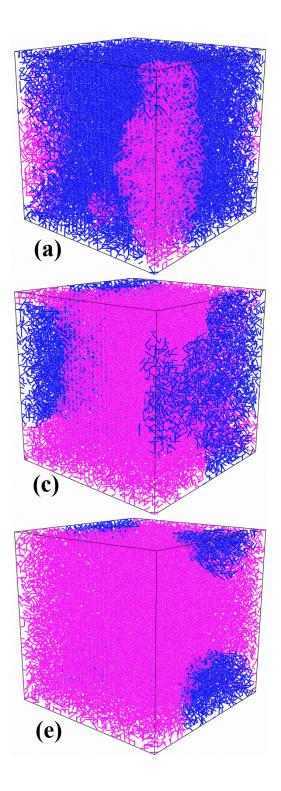
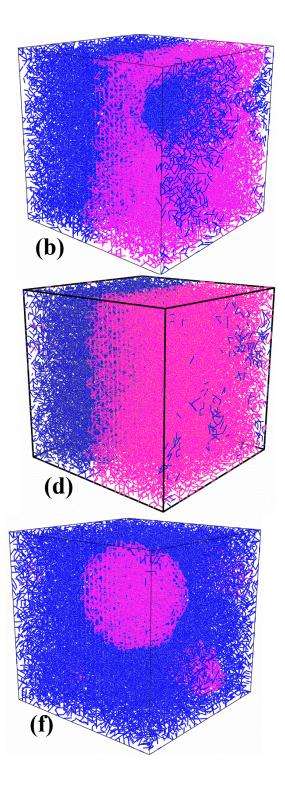
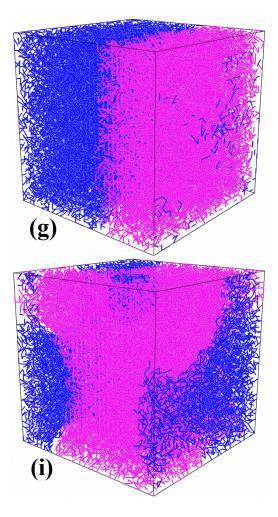


Figure S3. Snapshots of macrophase separated melt at  $U_p = 0.1$  during non-isothermal crystallization, for various compositions and  $\lambda$ : (a)  $\lambda = 1$ ,  $x_B = 0.125$ , (b)  $\lambda = 1$ ,  $x_B = 0.375$ , (c)  $\lambda = 1$ ,  $x_B = 0.5$ , (d)  $\lambda = 1$ ,  $x_B = 0.625$ , (e)  $\lambda = 1$ ,  $x_B = 0.875$ , (f)  $\lambda = 6$ ,  $x_B = 0.125$ , (g)  $\lambda = 6$ ,  $x_B = 0.375$ , (h)  $\lambda = 6$ ,  $x_B = 0.5$ , (i)  $\lambda = 6$ ,  $x_B = 0.625$  and (j)  $\lambda = 6$ ,  $x_B = 0.875$ . Blue and magenta colors represent bonds of A- and B-polymers, respectively.







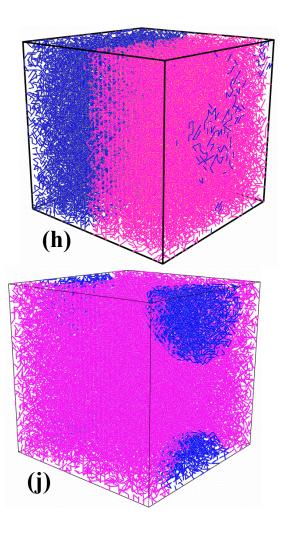
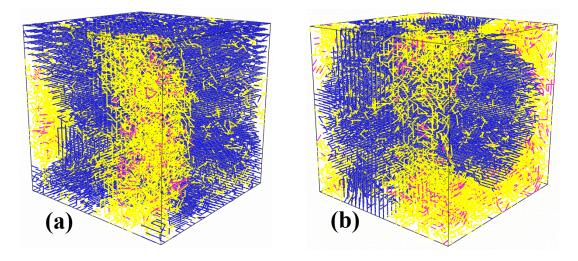
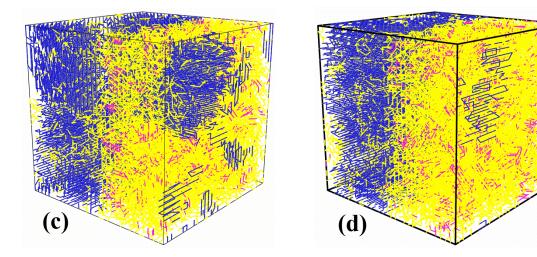
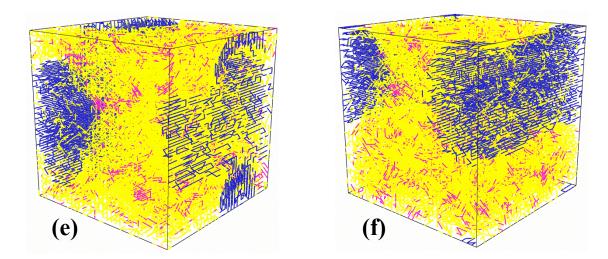


Figure S4. Snapshots of semicrystalline structure at  $U_p = 0.28$  during non-isothermal crystallization for various compositions at  $\lambda = 1$ : (a)  $x_B = 0.125$ , (b)  $x_B = 0.25$ , (c)  $x_B = 0.375$ , (d)  $x_B = 0.50$ , (e)  $x_B = 0.625$ , (f)  $x_B = 0.75$ , (g)  $x_B = 0.875$ . Blue and magenta colors represent crystalline bonds of A- and B-polymers, respectively. Yellow color represents non-crystalline bonds of both the polymers.







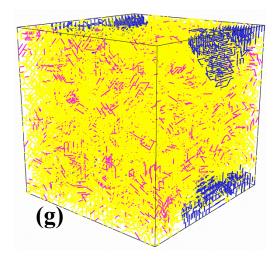


Figure S5. Change in crystallinity with  $U_p$  for A- and B-polymers: (a)  $X_A$  vs.  $U_p$  at  $\lambda = 1$ , (b)  $X_A$  vs.  $U_p$  at  $\lambda = 6$ , (c)  $X_B$  vs.  $U_p$  at  $\lambda = 1$ , (d)  $X_B$  vs.  $U_p$  at  $\lambda = 6$ .

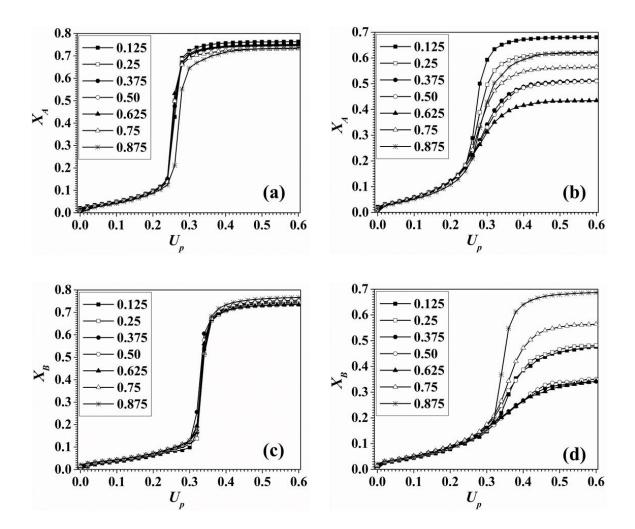
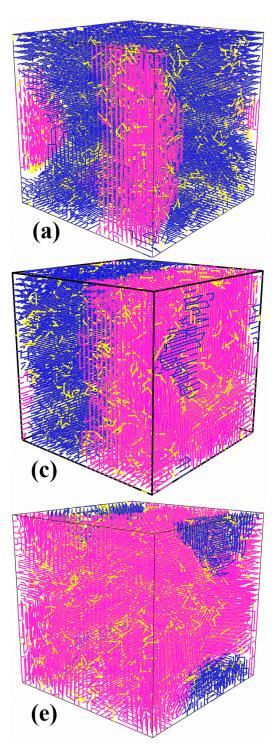
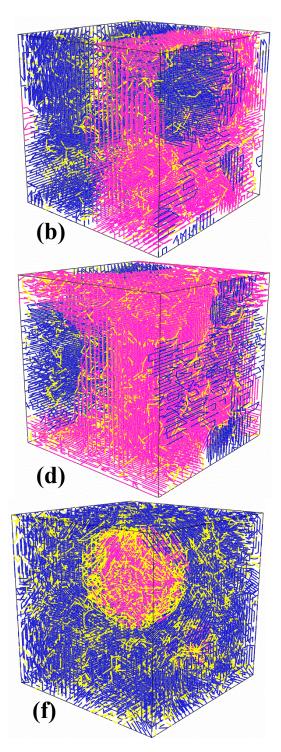
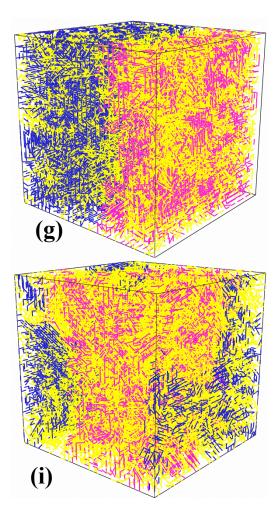


Figure S6. Snapshots of crystalline structure at  $U_p = 0.6$  during non-isothermal crystallization, for various compositions and  $\lambda$ : (a)  $\lambda = 1$ ,  $x_B = 0.125$ , (b)  $\lambda = 1$ ,  $x_B = 0.375$ , (c)  $\lambda = 1$ ,  $x_B = 0.5$ , (d)  $\lambda = 1$ ,  $x_B = 0.625$ , (e)  $\lambda = 1$ ,  $x_B = 0.875$ , (f)  $\lambda = 6$ ,  $x_B = 0.125$ , (g)  $\lambda = 6$ ,  $x_B = 0.375$ , (h)  $\lambda = 6$ ,  $x_B = 0.5$ , (i)  $\lambda = 6$ ,  $x_B = 0.625$  and (j)  $\lambda = 6$ ,  $x_B = 0.875$ . Blue and magenta colors represent crystalline bonds of A- and B-polymers, respectively. Yellow color represents non-crystalline bonds of both the polymers.







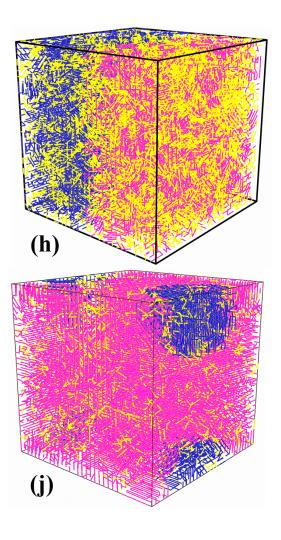


Figure S7. Change in average crystallite size with  $U_p$  for A- and B-polymers: (a)  $\langle S_A \rangle$  vs.  $U_p$  at  $\lambda = 1$ , (b)  $\langle S_A \rangle$  vs.  $U_p$  at  $\lambda = 6$ , (c)  $\langle S_B \rangle$  vs.  $U_p$  at  $\lambda = 1$ , (d)  $\langle S_B \rangle$  vs.  $U_p$  at  $\lambda = 6$ .

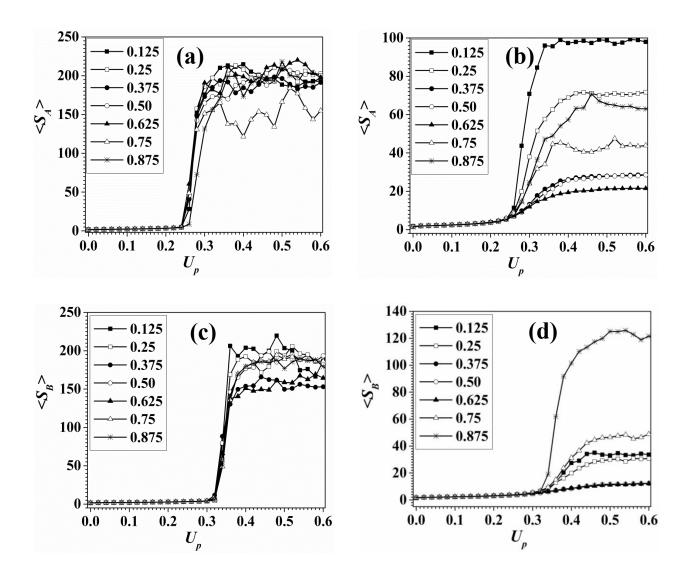
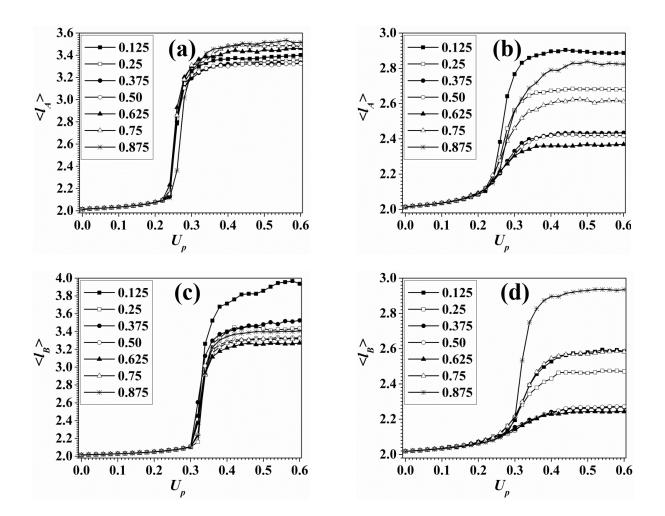


Figure S8. Change in average lamellar thickness with  $U_p$  for A- and B-polymers: (a)  $\langle l_A \rangle$  vs.  $U_p$  at  $\lambda = 1$ , (b)  $\langle l_A \rangle$  vs.  $U_p$  at  $\lambda = 6$ , (c)  $\langle l_B \rangle$  vs.  $U_p$  at  $\lambda = 1$ , (d)  $\langle l_B \rangle$  vs.  $U_p$  at  $\lambda = 6$ .



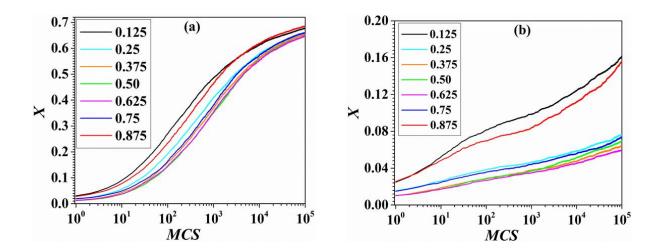


Figure S9. Change in isothermal overall crystallinity with number of Monte Carlo steps (*MCS*) at (a)  $\lambda = 1$  and (b)  $\lambda = 6$ .

	Weak segregation, $\lambda = 1$		Strong segregation, $\lambda = 6$	
Composition $(x_B)$	X <sub>A</sub>	X <sub>B</sub>	X <sub>A</sub>	X <sub>B</sub>
0.125	0.683	0.632	0.175	0.053
0.25	0.664	0.639	0.085	0.048
0.375	0.660	0.639	0.070	0.054
0.5	0.653	0.641	0.072	0.066
0.625	0.645	0.648	0.051	0.064
0.75	0.645	0.663	0.051	0.080
0.875	0.641	0.690	0.067	0.167

Table S1. Comparison in saturated crystallinity of A-polymer,  $X_A$ , B-polymer,  $X_B$ , with composition,  $x_B$ , during one-step isothermal crystallization, at  $\lambda = 1$  and 6.

	Two-step cooling		One-step cooling
<b>Composition</b> $(x_B)$	$U_{p} = 0.28$	$U_{p} = 0.6$	$U_p = 0.6$
0.125	3.47	3.51	2.7
0.25	3.46	3.46	2.67
0.375	3.39	3.44	2.68
0.5	3.43	3.47	2.74
0.625	3.45	3.47	2.73
0.75	3.53	3.63	2.81
0.875	3.61	3.72	2.95

Table S2. Comparison in average lamellar thickness of A-polymer,  $\langle l_A \rangle$ , with composition,  $x_B$ , during one- and two-step isothermal crystallization, at  $\lambda = 1$ .

Table S3. Comparison in average lamellar thickness of B-polymer,  $\langle l_B \rangle$ , with composition,  $x_B$ , during one- and two-step isothermal crystallization, at  $\lambda = 1$ .

	Two-step cooling	One-step cooling
<b>Composition</b> $(x_B)$	$U_p = 0.6$	$U_{p} = 0.6$
0.125	3.00	3.01
0.25	2.83	2.85
0.375	2.79	2.79
0.5	2.78	2.74
0.625	2.77	2.74
0.75	2.78	2.75
0.875	2.86	2.80

	Two-step cooling		One-step cooling	
<b>Composition</b> $(x_B)$	$U_{p} = 0.28$	$U_{p} = 0.6$	$U_{p} = 0.6$	
0.125	0.623	0.668	0.175	
0.25	0.397	0.473	0.085	
0.375	0.294	0.335	0.070	
0.5	0.384	0.426	0.072	
0.625	0.358	0.422	0.051	
0.75	0.276	0.329	0.051	
0.875	0.429	0.523	0.067	

Table S4. Comparison in saturated crystallinity of A-polymer,  $X_A$ , with composition,  $x_B$ , during one- and two-step isothermal crystallization, at  $\lambda = 6$ .

Table S5. Comparison in saturated crystallinity of B-polymer,  $X_B$ , with composition,  $x_B$ , during one- and two-step isothermal crystallization, at  $\lambda = 6$ .

	Two-step cooling		One-step cooling
<b>Composition</b> $(x_B)$	$U_{p} = 0.28$	$U_p = 0.6$	$U_p = 0.6$
0.125	0.126	0.342	0.053
0.25	0.144	0.243	0.048
0.375	0.145	0.180	0.054
0.5	0.151	0.250	0.066
0.625	0.143	0.259	0.064
0.75	0.150	0.246	0.051
0.875	0.137	0.523	0.167