## Supporting Information

# Effect of Composition Asymmetry on the Phase Separation and Crystallization in Double Crystalline Binary Polymer Blends: A Dynamic Monte Carlo Simulation Study 

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Figure S1. Snapshots at $U_{p}=0$ for various compositions, $x_{B}$ : (a) 0.125 , (b) 0.375, (c) 0.5, (d) 0.625 and (e) 0.875. Blue and magenta colors represent bonds of A- and B-polymers, respectively.


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Figure S2. Change in mean square displacement of center of mass ( $d_{c m}^{2}$ ) with $U_{p}$ for Aand B-polymers: (a) $d_{c m_{-} A}^{2}$ vs. $U_{p}$ at $\lambda=\mathbf{1}$, (b) $d_{c m_{-} A}^{2}$ vs. $U_{p}$ at $\lambda=6$, (c) $d_{c m_{-} B}^{2}$ vs. $U_{p}$ at $\lambda$ $=\mathbf{1}$, (d) $d_{c m_{-} B}^{2}$ vs. $U_{p}$ at $\lambda=\mathbf{6}$.





Figure S3. Snapshots of macrophase separated melt at $U_{p}=\mathbf{0 . 1}$ during non-isothermal crystallization, for various compositions and $\lambda$ : (a) $\lambda=1, x_{B}=0.125$, (b) $\lambda=1, x_{B}=0.375$, (c) $\lambda=1, x_{B}=0.5$, (d) $\lambda=1, x_{B}=\mathbf{0 . 6 2 5}$, (e) $\lambda=1, x_{B}=0.875$, (f) $\lambda=6, x_{B}=0.125$, (g) $\lambda=$ 6, $x_{B}=0.375$, (h) $\lambda=6, x_{B}=0.5$, (i) $\lambda=6, x_{B}=0.625$ and (j) $\lambda=6, x_{B}=0.875$. Blue and magenta colors represent bonds of A- and B-polymers, respectively.




Figure S4. Snapshots of semicrystalline structure at $U_{p}=0.28$ during non-isothermal crystallization for various compositions at $\lambda=1$ : (a) $x_{B}=0.125$, (b) $x_{B}=0.25$, (c) $x_{B}=$ 0.375 , (d) $x_{B}=0.50$, ( $\left.\mathbf{e}\right) x_{B}=0.625$, (f) $x_{B}=0.75$, (g) $x_{B}=0.875$. Blue and magenta colors represent crystalline bonds of A- and B-polymers, respectively. Yellow color represents non-crystalline bonds of both the polymers.



Figure S5. Change in crystallinity with $U_{p}$ for A- and B-polymers: (a) $X_{A}$ vs. $U_{p}$ at $\lambda=$ 1, (b) $X_{A}$ vs. $U_{p}$ at $\lambda=6$, (c) $X_{B}$ vs. $U_{p}$ at $\lambda=1$, (d) $X_{B}$ vs. $U_{p}$ at $\lambda=6$.





Figure S6. Snapshots of crystalline structure at $U_{p}=0.6$ during non-isothermal crystallization, for various compositions and $\lambda$ : (a) $\lambda=1, x_{B}=0.125$, (b) $\lambda=1, x_{B}=0.375$, (c) $\lambda=1, x_{B}=0.5$, (d) $\lambda=1, x_{B}=\mathbf{0 . 6 2 5}$, (e) $\lambda=1, x_{B}=0.875$, (f) $\lambda=6, x_{B}=0.125$, (g) $\lambda=$ 6, $x_{B}=\mathbf{0 . 3 7 5}$, (h) $\lambda=6, x_{B}=\mathbf{0 . 5}$, (i) $\lambda=6, x_{B}=0.625$ and (j) $\lambda=6, x_{B}=0.875$. Blue and magenta colors represent crystalline bonds of A- and B-polymers, respectively. Yellow color represents non-crystalline bonds of both the polymers.




Figure S7. Change in average crystallite size with $U_{p}$ for A- and B-polymers: (a) $\left\langle S_{A}\right\rangle$ vs. $U_{p}$ at $\lambda=1$, (b) $\left\langle S_{A}\right\rangle$ vs. $U_{p}$ at $\lambda=6$, (c) $\left\langle S_{B}\right\rangle$ vs. $U_{p}$ at $\lambda=1$, (d) $\left\langle S_{B}\right\rangle$ vs. $U_{p}$ at $\lambda=6$.





Figure S8. Change in average lamellar thickness with $U_{p}$ for A- and B-polymers: (a) $\left\langle l_{A}\right\rangle$ vs. $U_{p}$ at $\lambda=1$, (b) $\left\langle l_{A}\right\rangle$ vs. $U_{p}$ at $\lambda=6$, (c) $\left\langle l_{B}\right\rangle$ vs. $U_{p}$ at $\lambda=1$, (d) $\left\langle l_{B}\right\rangle$ vs. $U_{p}$ at $\lambda=6$.





Figure S9. Change in isothermal overall crystallinity with number of Monte Carlo steps (MCS) at (a) $\lambda=1$ and (b) $\lambda=6$.



Table S1. Comparison in saturated crystallinity of A-polymer, $X_{A}$, B-polymer, $X_{B}$, with composition, $x_{B}$, during one-step isothermal crystallization, at $\lambda=1$ and 6 .

|  | Weak segregation, $\lambda=1$ |  | Strong segregation, $\lambda=6$ |  |
| :---: | :---: | :---: | :---: | :---: |
| Composition <br> $\left(x_{B}\right)$ | $X_{A}$ | $X_{B}$ | $X_{A}$ | $X_{B}$ |
| 0.125 | 0.683 | 0.632 | 0.175 | 0.053 |
| 0.25 | 0.664 | 0.639 | 0.085 | 0.048 |
| 0.375 | 0.660 | 0.639 | 0.070 | 0.054 |
| 0.5 | 0.653 | 0.641 | 0.072 | 0.066 |
| 0.625 | 0.645 | 0.648 | 0.051 | 0.064 |
| 0.75 | 0.645 | 0.663 | 0.051 | 0.080 |
| 0.875 | 0.641 | 0.690 | 0.067 | 0.167 |

Table S2. Comparison in average lamellar thickness of A-polymer, $\left\langle l_{A}\right\rangle$, with composition, $x_{B}$, during one- and two-step isothermal crystallization, at $\lambda=1$.

|  | Two-step cooling |  | One-step cooling |
| :---: | :---: | :---: | :---: |
| Composition $\left(x_{B}\right)$ | $U_{p}=0.28$ | $U_{p}=0.6$ | $U_{p}=0.6$ |
| 0.125 | 3.47 | 3.51 | 2.7 |
| 0.25 | 3.46 | 3.46 | 2.67 |
| 0.375 | 3.39 | 3.44 | 2.68 |
| 0.5 | 3.43 | 3.47 | 2.74 |
| 0.625 | 3.45 | 3.47 | 2.73 |
| 0.75 | 3.53 | 3.63 | 2.81 |
| 0.875 | 3.61 | 3.72 | 2.95 |

Table S3. Comparison in average lamellar thickness of B-polymer, $\left\langle l_{B}\right\rangle$, with composition, $x_{B}$, during one- and two-step isothermal crystallization, at $\lambda=1$.

|  | Two-step cooling | One-step cooling |
| :---: | :---: | :---: |
| Composition $\left(x_{B}\right)$ | $U_{p}=0.6$ | $U_{p}=0.6$ |
| 0.125 | 3.00 | 3.01 |
| 0.25 | 2.83 | 2.85 |
| 0.375 | 2.79 | 2.79 |
| 0.5 | 2.78 | 2.74 |
| 0.625 | 2.77 | 2.74 |
| 0.75 | 2.78 | 2.75 |
| 0.875 | 2.86 | 2.80 |

Table S4. Comparison in saturated crystallinity of A-polymer, $X_{A}$, with composition, $x_{B}$ , during one- and two-step isothermal crystallization, at $\lambda=6$.

|  | Two-step cooling |  | One-step cooling |
| :---: | :---: | :---: | :---: |
| Composition $\left(x_{B}\right)$ | $U_{p}=0.28$ | $U_{p}=0.6$ | $U_{p}=0.6$ |
| 0.125 | 0.623 | 0.668 | 0.175 |
| 0.25 | 0.397 | 0.473 | 0.085 |
| 0.375 | 0.294 | 0.335 | 0.070 |
| 0.5 | 0.384 | 0.426 | 0.072 |
| 0.625 | 0.358 | 0.422 | 0.051 |
| 0.75 | 0.276 | 0.329 | 0.051 |
| 0.875 | 0.429 | 0.523 | 0.067 |

Table $\mathbf{S 5}$. Comparison in saturated crystallinity of B-polymer, $X_{B}$, with composition, $x_{B}$ , during one- and two-step isothermal crystallization, at $\lambda=6$.

|  | Two-step cooling |  | One-step cooling |
| :---: | :---: | :---: | :---: |
| Composition $\left(x_{B}\right)$ | $U_{p}=0.28$ | $U_{p}=0.6$ | $U_{p}=0.6$ |
| 0.125 | 0.126 | 0.342 | 0.053 |
| 0.25 | 0.144 | 0.243 | 0.048 |
| 0.375 | 0.145 | 0.180 | 0.054 |
| 0.5 | 0.151 | 0.250 | 0.066 |
| 0.625 | 0.143 | 0.259 | 0.064 |
| 0.75 | 0.150 | 0.246 | 0.051 |
| 0.875 | 0.137 | 0.523 | 0.167 |


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