Shortlisting of Composite CaO-Based Structured Bodies Suitable For Thermochemical Heat Storage With The CaO/Ca(OH)₂ Reaction Scheme

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The file includes: Composite CaCO₃/kaolinite pellets Estimation of free CaO content in composite materials XRD patterns of composite CaCO₃/kaolinite pellets Inertness of Ca₂Al₂SiO₇ towards hydration reaction Hydration capacity and rate of composite pellet and powder Figures S1, S2, S3, S4 and S5 Table S1

Composite CaCO₃/kaolinite pellets

Emphasizing on the mechanical properties of composite CaO/kaolinite pellets, preliminary evaluation experiments were carried out on pellets with kaolinite content of 10 and 25 wt %. For the preparation of composite pellets, Natural89 (\geq 89 wt % CaCO₃) and Natural98 (\geq 98.7 wt % CaCO₃) were employed as Ca-precursors, with Natural89 powder being used after calcination at 900°C for 2 h under air followed by storage at ambient atmosphere, therefore generating Ca(OH)₂. The manufacturing process included Ca-precursor and kaolinite mixing via the solid mixing technique (following the rest of steps as described in detail in the manuscript) and stabilization of the final pellets via calcination at 900 °C for 4 h. In Natural89 derived materials, XRD analysis on milled pellets right after their calcination revealed the formation of the ternary Ca/Al/Si phase regardless of kaolinite content applied. However, examining the mechanical properties of the fresh calcined pellets, it was found that formulations with 10 wt % of kaolinite exhibited low mechanical strength values (between 2 and 7 N), whereas pellets with higher kaolinite content proved to be less susceptible to breakage with CS values of 16-19 N. Such CS measurements were carried out on multiple pellets (around 10 per sample) and values presented here correspond to the range noted. When Natural98 was employed as Ca-precursor, the composition was similar to the one identified in Natural89 materials including CaO, Ca₂Al₂SiO₇ and SiO₂ peaks. Regarding mechanical stability, Natural98_kaolinite25 pellets presented higher resistance to breakage with CS values of 22-30 N.

Т. н. н. / С	Composition	CS _{fresh} / N
I calcination / oC		
900	CaO, Ca ₂ Al ₂ SiO ₇ , SiO ₂	2-7
900	CaO, Ca ₂ Al ₂ SiO ₇ , SiO ₂	16-19
900	CaO, Ca ₂ Al ₂ SiO ₇ , SiO ₂	22-30
	900	Tcalcination / oC CaO, Ca ₂ Al ₂ SiO ₇ , SiO ₂ 900 CaO, Ca ₂ Al ₂ SiO ₇ , SiO ₂ 900 CaO, Ca ₂ Al ₂ SiO ₇ , SiO ₂

Table S1. List of materials examined: Composition and mechanical properties of fresh pellets

Estimation of free CaO content in composite materials

The estimation of free CaO content after calcination was based on the reasoning/assumption that during calcination and apart from the solid state reaction of the two precursors (eq. 3 in the manuscript), two side reactions take place, including the decarbonation of CaCO₃ resulting to CaO (eq. S1) and the dehydration of Al₂Si₂O₅(OH)₄ resulting to Al₂SiO₇ (eq. S2). The decarbonation reaction could be considered as competitive. From the literature, though, it is known that decarbonation can initiate from 650 °C¹. Moreover, according to XRD analysis performed on CaCO₃/kaolinite samples calcined at various temperatures between 750 and 1000 °C (not here presented), Ca₂Al₂SiO₇ formation is favored at T > 800 °C. Thus, composite CaO/Ca₂Al₂SiO₇ materials production would not be inhibited. The purity of Ca-precursor was crucial and it was assumed that the impurities contained in CaCO₃ grades were completely inert during both gehlenite formation and cyclic reactions. Considering the weight ratio of the two precursors, the stoichiometric formation of Ca₂Al₂SiO₇ and the side products according to eq. 3 (in the manuscript), free CaO was calculated as the difference between the total CaO employed and the CaO consumed for Ca₂Al₂SiO₇ formation.

$$\begin{array}{ccc} \xrightarrow{} & & & \\ CaCO_{3(s)} & & & \\ (S1) & & & \\ \end{array} + & & CO_{2(g)} \end{array}$$

$$Al_2Si_2O_5(OH)_{4(s)} \rightarrow Al_2SiO_{7(s)} + 2H_2O_{(g)}$$
 (S2)

In particular, an example for Natural98 is provided and the following procedure was followed:

- It is assumed that all impurities remain inert during calcination/hydration/carbonation.
- 1g of Natural98 limestone (98.7 % purity) renders 0.553 g of CaO through calcination (eq. S1), with 1.3% of impurities remaining inert. (MW _{CaO}=56.08 g/mol, MW _{CaCO3} = 100.09 g/mol) Thus, in 75% wt of Natural98 limestone 0.415 g of CaO (74% wt due to impurities) are contained.

¹ Halikia, I.; Zoumpoulakis, L.; Christodoulou, E.; Prattis, D. *The European Journal of Mineral Processing and Environmental Protection* **2001**, 1, 2, 89-102

- The dehydration of 25% wt of $Al_2Si_2O_5(OH)_4$ (eq. S2) renders 0.215 g of Al_2SiO_7 (MW Al2Si2O5(OH)4 = 258.16 g/mol, MW Al2Si2O7 = 222.13 g/mol)
- Therefore, the reaction of 0.215 g of Al₂SiO₇ requires 0.109 g of CaO (eq. 3 in the manuscript) for gehlenite formation.
- Concluding, free CaO is equal to 0.415-0.109=0.306 g of CaO, corresponding to 49% of the total amount of material produced (including Ca₂Al₂SiO₇ and SiO₂).

Similarly, free CaO content can be estimated for Natural89 as 45% of final material.

XRD patterns of composite CaCO₃/kaolinite pellets

Figures S1, S2 and S3 present the results from the XRD analysis performed on the composite materials studied in the current work in their fresh form (i.e. after calcination). In these comparative diagrams, the reference diffraction peaks of the three main crystal phases –CaO, Ca₂Al₂SiO₇ and SiO₂- are depicted with dotted lines. Evidently, the main crystal phase identified in all composite materials is CaO, assigned to the [200] crystallographic plane (2theta=37.36 degrees). The addition of kaolinite resulted in the formation of a mixed Ca/Al/Si phase corresponding to gehlenite (Ca₂Al₂SiO₇). The main peak of the ternary Ca/Al/Si phase (Ca₂Al₂SiO₇) characterized the [211] crystallographic plane (2theta=31.42 degrees) and it was profound in almost all CaCO₃/kaolinite composite materials. Also, SiO₂ phase is evident in all materials, mainly the one assigned to the [011] crystallographic plane (2theta=26.64 degrees).

Regarding Figure S1, notably, increasing kaolinite amount in the composite materials did not affect the intensity of $Ca_2Al_2SiO_7$ peaks. However, it resulted in SiO_2 peaks with higher intensity in material Kaol70 than in the rest of compositions. Given that pure kaolinite formulations (denoted as Kaol100) consisted of almost single-phase SiO_2 peaks, such a decrease in their intensity in composite materials is natural considering the addition/increase in CaO content employed. Therefore, gradually lower amounts of kaolinite were employed for Kaol70 to Kaol25 leading to SiO₂ peaks with lower intensity.

In relation to samples presented in Figure S2, as it is also stated in the main manuscript, employing $CaCO_3$ precursors with different particle size affected $Ca_2Al_2SiO_7$ peaks. In particular, in Natural98 derived materials with smaller $CaCO_3$ particle sizes, $Ca_2Al_2SiO_7$ peak became more intense and narrower, resulting to different crystallite sizes (as shown in Table 5 in the manuscript). Natural89 derived sample presented similar behaviour with the corresponding Natural98derived one.

As depicted in Figure S3 and also stated in the main manuscript, the application of different conditions during preparation of the Natural98-derived composite materials did not affect notably the morphology of the crystal phases peaks. On the other hand, in the case of Natural89-derived samples, it is clear that calcining the material for 2 h resulted in no formation of the ternary Ca/Al/Si phase, while there were clear Ca(OH)₂ peaks at 18, 34, 47 and 51 degrees.

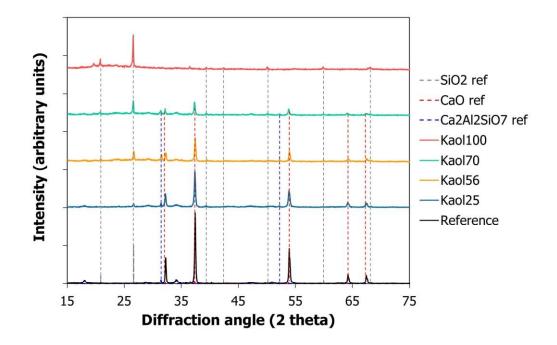


Figure S1. XRD analysis for Reference and clay composite materials (JCPDS Card Numbers used: CaO: 00-048-1467; Ca₂Al₂SiO₇: 00-035-0755; SiO₂: 00-085-0695)

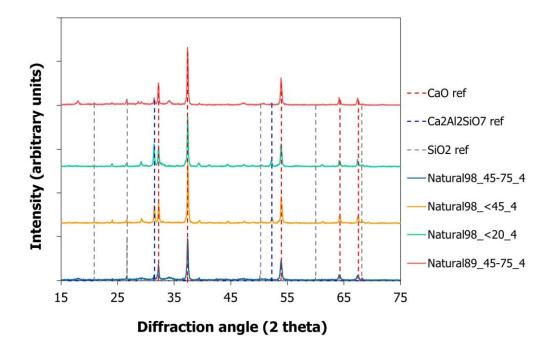


Figure S2. XRD analysis for composite CaO/kaolinite (75/25 %wt) materials with different Caprecursors (JCPDS Card Numbers used: CaO: 00-048-1467; Ca₂Al₂SiO₇: 00-035-0755; SiO₂: 00-0850695)

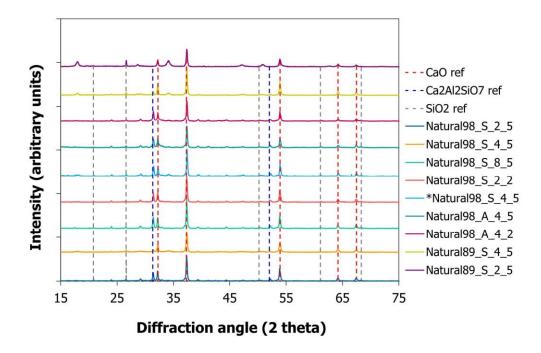


Figure S3. XRD analysis for composite CaO/kaolinite (75/25 %wt) materials prepared under different conditions (JCPDS Card Numbers: CaO: 00-048-1467; Ca₂Al₂SiO₇: 00-035-0755; SiO₂: 00-085-0695)

Inertness of Ca₂Al₂SiO₇ towards hydration reaction

Considering the difficulty to obtain pure Ca₂Al₂SiO₇, another approach was followed in order to define the participation of the ternary Ca/Al/Si phase in the hydration reaction. To this respect, multiple composite pellets of Natural98 derived material were subjected to hydration according to the protocol described in the main document and the particles were obtained at their hydrated form. XRD analysis was conducted on these pellets in powder form (right after hydration). According to results obtained (Figure S4), only Ca(OH)₂ peaks were identified, thereby proving that the Ca₂Al₂SiO₇ phase is not subjected to hydration under the conditions applied in the present study.

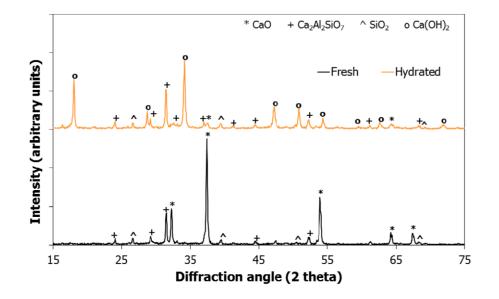


Figure S4. XRD analysis of fresh and hydrated Natural98 derived pellets

Hydration capacity and rate of composite pellet and powder

On an effort to determine if the structured shape of the composite materials examined in this study affects the hydration capacity and reaction rate of the materials, preliminary experiments were conducted on a composition in pellet and powder form. The material studied derived from Natural98 CaCO₃ particles with a size <45 µm combined with 25 wt % of kaolinite and was calcined at 900 °C for 4 h with 5 °C·min⁻¹. A 5-cycle experiment was performed both in pellet and powder form (amount equal to 1 milled formulation). The evaluation protocol applied in this experiment included hydration at 450°C for 15min and dehydration at 550 °C for 20 min, while the atmosphere applied in both reactions was 75 vol % of steam (P_{total}=1 bar). Figure S5 shows the hydration capacity measured for both samples on the basis of total fresh material mass (after calcination), as well as the hydration rate noted for these samples for cycles No. 3 and 5. As it can be seen in Figure S5a, the hydration capacity noted for both samples was similar and around 30% w.r.t. the maximum theoretical value. These capacities are slightly lower than the one presented for Natural98 <45 pellet in Table 6 of the manuscript corresponding to a hydration efficiency of 40%, which is justified by the lower steam concentration applied in this preliminary examination. Moreover, the analysis of both measurements in Figure S5b showed that the reaction rates between the pellet and powder samples were very close. In particular, for the composite pellet, hydration reaction at cycle 3 began by t=50 s, while for the powder it began at about 80 s. However, they became identical by t=170 s during the fast part of the reaction and were completed by t=600 s. In cycle 5, hydration reaction started at the same time for both samples, but hydration conversion was achieved slightly faster for the composite powder reaching its maximum value at t=500 s. Therefore, the structured shape of the composite materials did not have any notable effect on materials hydration performance.

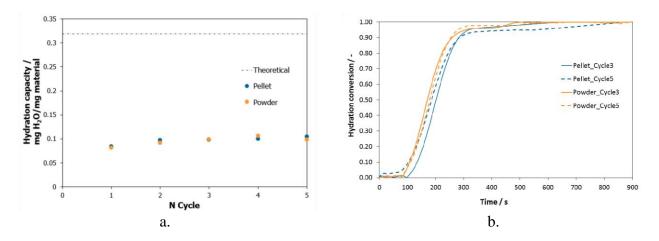


Figure S5. a. Hydration performance of composite pellet and powder in the course of 5 hydration/dehydration cycles, b. Hydration conversion of composite pellet and powder with time