

Supplementary Information:

**Atomistic Observation of Structural Evolution during Magnesium
Oxide Growth**

Fan Cao[†], He Zheng[†], Shuangfeng Jia^{,†}, Huihui Liu[†], Lei Li[†], Boyun Chen^{†,‡}, Xi Liu^{†,§}, Shujing
Wu[†], Huaping Sheng[†], Ru Xing[⊥], Dongshan Zhao[†], and Jianbo Wang^{*,†}*

*[†] School of Physics and Technology, Center for Electron Microscopy, MOE Key Laboratory of
Artificial Micro- and Nano-structures, and Institute for Advanced Studies, Wuhan University,
Wuhan 430072, China*

[‡] Shuiguohu Senior Middle School, Wuhan 430071, China

*[§] Middle School Attached to Huazhong University of Science and Technology, Wuhan 430074,
China*

*[⊥] The Department of Physics Science and Technology, Baotou Normal College, Baotou 014030,
China*

Movie 1. Real-time video showing the growth of the three MgO variants. The time-code is shown in the format of mm:ss:ff (minutes:seconds:frame-intervals).

Movie 2. Atomic-scale observation of the grain rotation and GB migration process. The time-code is shown in the format of mm:ss:ff (minutes:seconds:frame-intervals).

Movie 3. The dislocation-mediated GB migration, as cut out from Movie 2. The time-code is shown in the format of mm:ss:ff (minutes:seconds:frame-intervals), and the time for beginning in Movie 3 is corresponding to 07:14:30 in Movie 2.

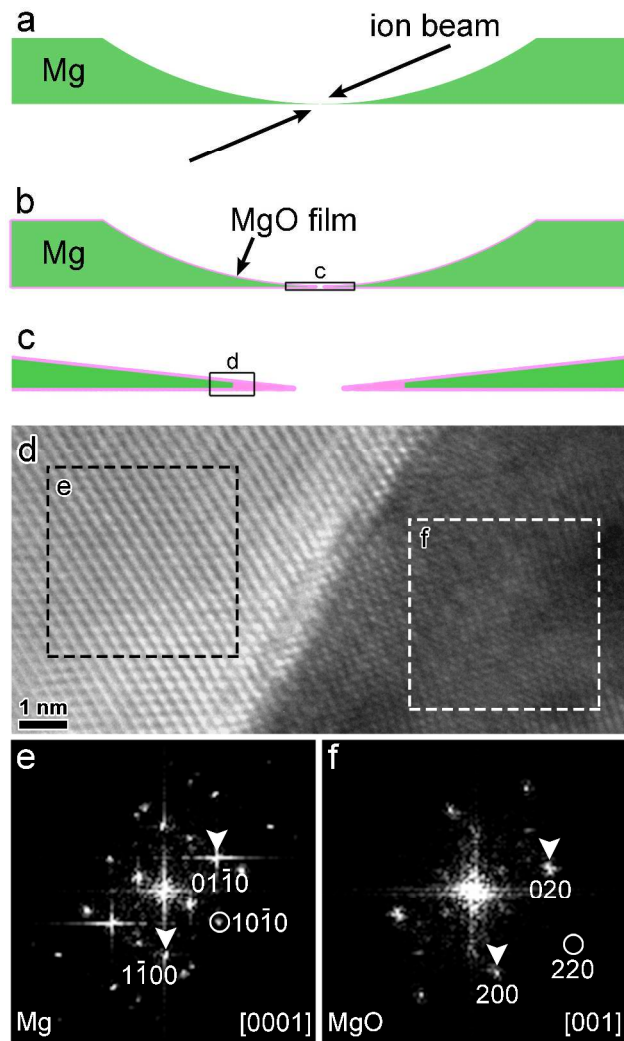


Figure S1. a) Cross-section view of obtaining a thin Mg substrate (colored green) by ion polishing. b) After annealed in air, the substrate was covered with MgO (colored pink). c) The thin area that could be penetrated by e-beam, corresponding to the black-boxed region in (b). d) HRTEM image corresponding to the black-boxed region in (c). e,f) Fast Fourier transformation (FFT) patterns corresponding to the left and right part of (d), respectively, implying the OR: $[0001]_{\text{H}} // [001]_{\text{C}}$, $(10\bar{1}0)_{\text{H}} // (110)_{\text{C}}$.

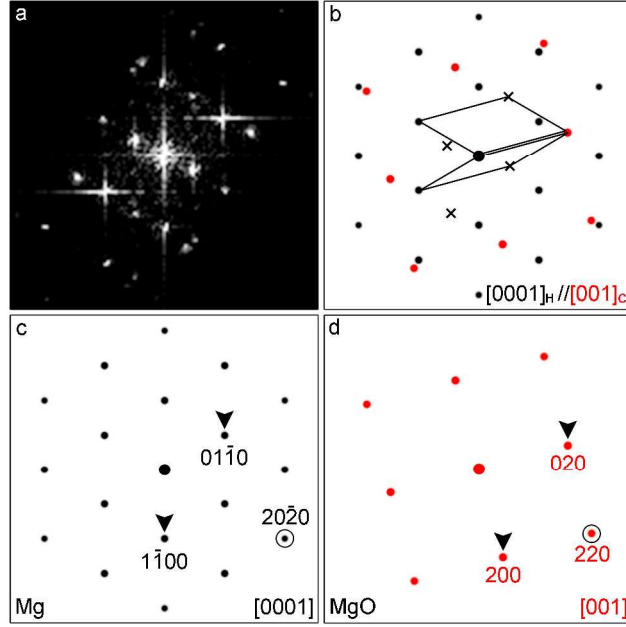


Figure S2. Further analysis of Figure S1e. a) The same FFT pattern as shown in S1e. b) Corresponding simulated diffraction pattern, comprised of (c) and (d), which indicates the OR: $[0001]_H // [001]_C$, $(10\bar{1}0)_H // (110)_C$ between Mg and MgO. The symbol “x” represents double diffraction spots and the routes are pointed out by black lines. c,d) Simulated diffraction patterns of Mg along $[0001]$ and MgO along $[001]$ directions, respectively.

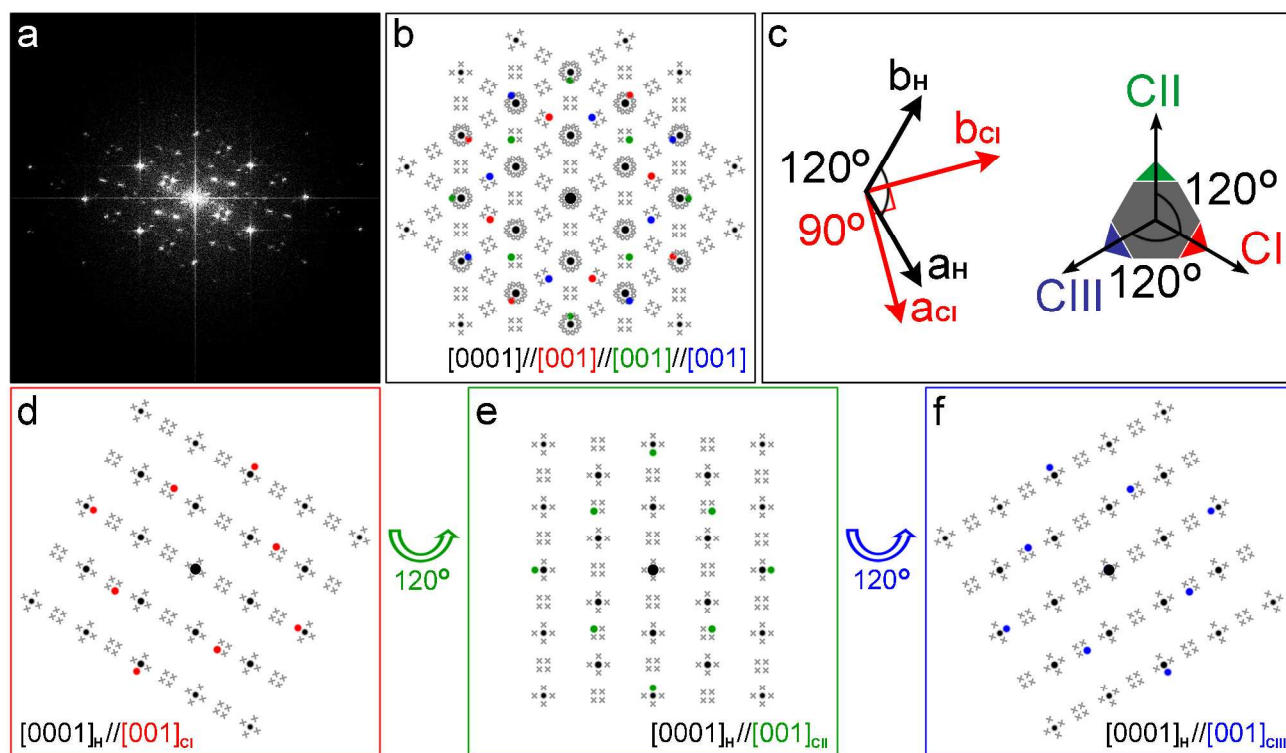


Figure S3. a,b) Fast Fourier-transformed (FFT) pattern of Mg substrate covered by MgO and the corresponding simulated diffraction pattern based on d-f): Mg matrix and the three MgO variants (“x” represents double diffraction spots), indicating the existence of three MgO variants. c) Schematic illustration of the OR between Mg matrix and the three MgO variants.

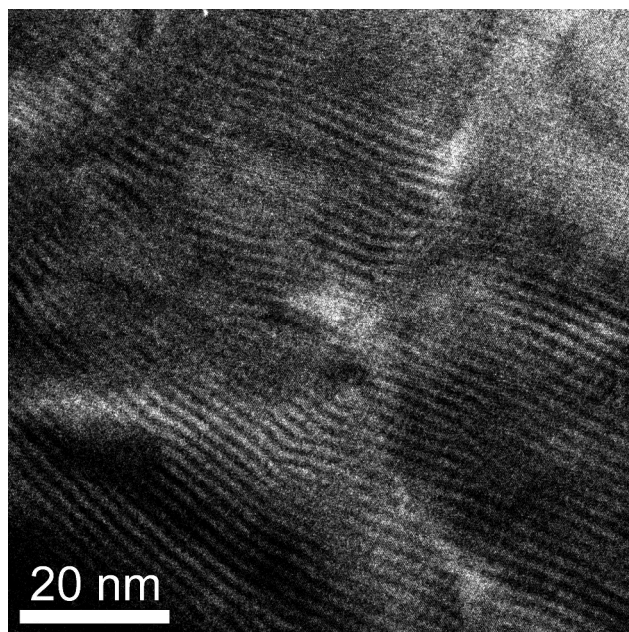


Figure S4. A lower magnification TEM image showing the Moiré fringes mentioned in Figure 1b.

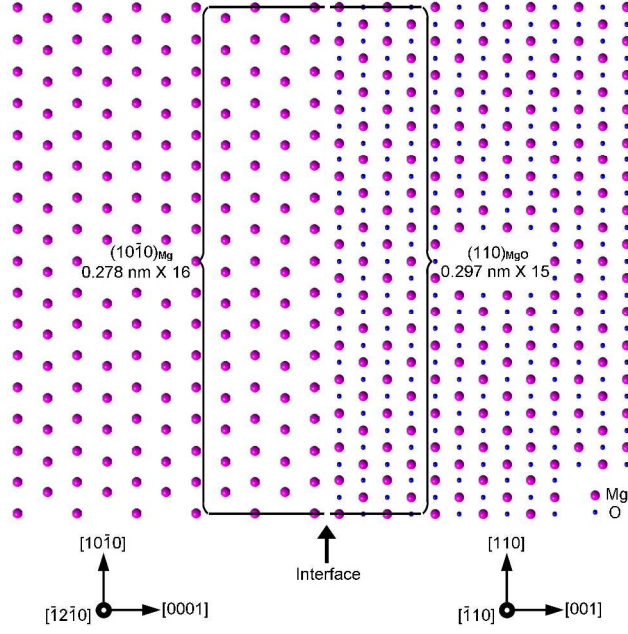


Figure S5. Side view of the possible atomic arrangements at the interface of Mg and MgO, based on a near coincidence site lattice (CSL) model. The mismatch is estimated to be $(0.297 \times 15 - 0.278 \times 16) / (0.278 \times 16) = 0.16\%$.

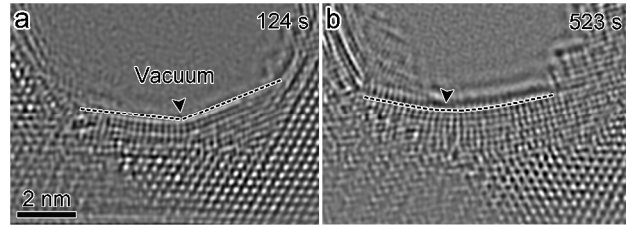


Figure S6. a,b) Same images shown in Figure 3d and e in the manuscript, respectively, depicting the GB between MgO grains and the vacuum with dashed black lines.

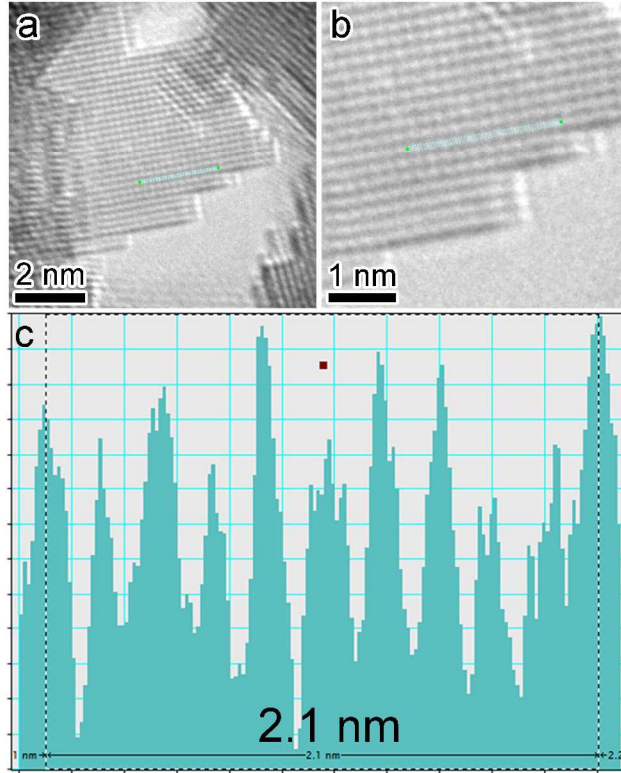


Figure S7. a) A dislocation-free MgO lattice image is chosen as the reference for the measurement of lattice strain. b-c) The spacing of $\{200\}_C$ planes is measured to be 0.21 nm, which matches well with the theoretical value (0.21 nm). As a result, the reference of sextuple lattice distance is $D = 0.21 \text{ nm}$.