Buffer Layer Assisted Epitaxial Growth of Perfectly Aligned Oxide Nanorod Arrays in Solution

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- 1. Additional SEM images of the ZnO NRs synthesized on the $ZnAl_2O_4$ epilayer/ α -sapphire substrates
- 2. Fitting diameters and heights of NRs as functions of their density
- 3. SEM images of ZnO NRs grown on the α -sapphire substrates partially covered with a thick ZnAl₂O₄ layer
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- 6. Growth behaviors of ZnO NRs on the MgAl₂O₄ epilayer/ α -sapphire substrates

1. Additional SEM images of the ZnO NRs synthesized on the ZnAl₂O₄ epilayer/ α -sapphire

substrates

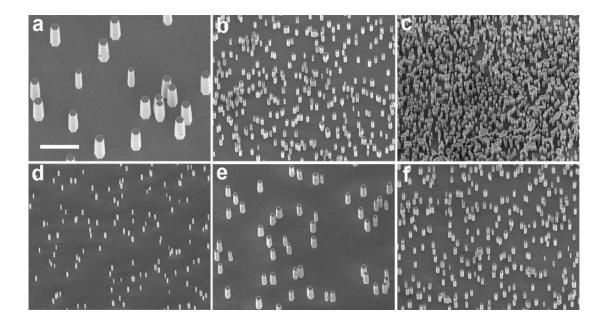


Figure S1. SEM images of the ZnO NRs grown on the $ZnAl_2O_4/\alpha$ -sapphire substrates which were formed by annealing at (a) 800 °C, (b) 900 °C and (c) 1000 °C, with $[Zn(NO_3)_2]=20$ mM. SEM images of the ZnO NRs grown on the $ZnAl_2O_4/\alpha$ -sapphire substrates which were prepared at 900 °C, with $[Zn(NO_3)_2]$: (d) 5 mM (e) 10 mM and (f) 20 mM. All the figures have the same scale bar of 10 μ m.

2. Fitting diameters and heights of NRs as functions of their density

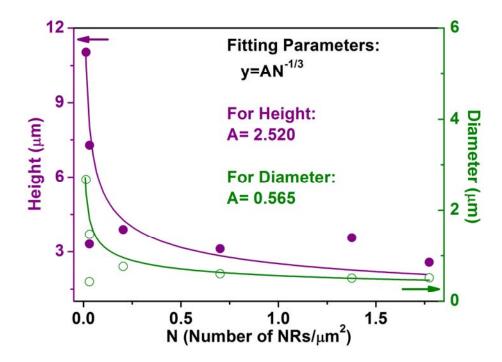


Figure S2. Fitting diameters and heights of NRs to a function that $y = \frac{A}{\sqrt[3]{N}}$. The constant A has a dimension of $\mu m^{1/3}$.

3. SEM images of ZnO NRs grown on the α-sapphire substrates partially covered with a thick

ZnAl₂O₄ layer

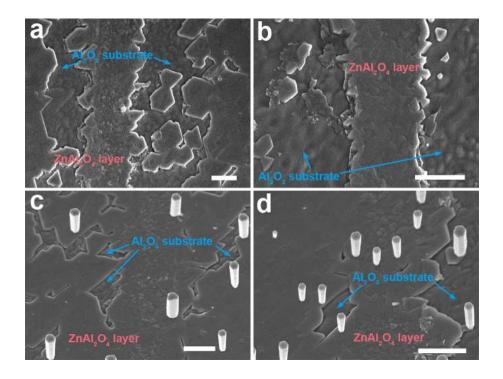


Figure S3. SEM images of the $ZnAl_2O_4$ layer on the α -sapphire substrate formed by annealing at 1000 °C before (a, b) and after (c, d) growing the ZnO NRs. The scale bars represent 1 μ m. It is clear that the ZnO NRs can only grow on ZnAl₂O₄ covered regions, whereas the exposed sapphire surface cannot initiate any NR nucleation.

4. Growth behaviors of ZnO NRs on the ZnAl₂O₄/ (0001) sapphire (c plane) substrates

For the (0001) sapphire (c plane) substrate, previous studies demonstrated that epitaxial ZnAl₂O₄ layer can also be generated if the annealing temperature is higher than 850 °C,^{S1, S2} which is consistent with our experiments. From the XRD measurements, we found that ZnAl₂O₄ (111), (222) and (333) are the most intense peaks (Figure S4b). However, the presence of other strong ZnAl₂O₄ peaks implies a poor epitaxial relationship at the interface. Moreover, the surface rms roughness of the ZnAl₂O₄ layer is much larger than that formed on α -sapphire at the same temperature (Figure S4a). These factors result in the NRs grown on the ZnAl₂O₄/c-sapphire substrate showing poorer orientation and density controls than the ZnAl₂O₄/ α -sapphire counterpart (Figure S4c-e).

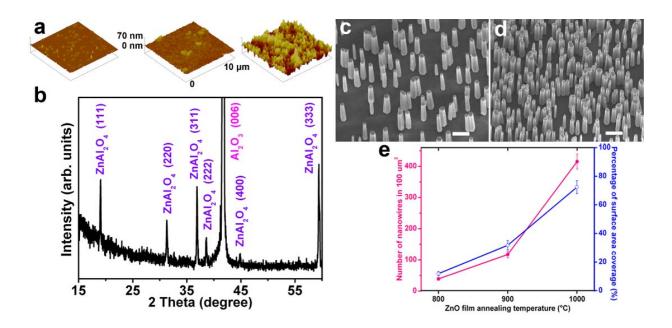


Figure S4. (a). AFM micrographs of the ZnAl₂O₄/c-sapphire substrates created by annealing at 800, 900 and 1000 °C, the rms roughnesses are 2.415 nm, 4.204 nm, 19.365 nm, respectively. (b). XRD spectra of the ZnAl₂O₄/c-sapphire substrate formed at 1000 °C. SEM images of ZnO NRs grown on the ZnAl₂O₄/c-sapphire substrates which were created by annealing at (c) 800 °C and (d) 900 °C, [Zn(NO₃)₂]= 20 mM. The scale bars represent 2 μ m. (e). Dependences of density and surface area coverage on the ZnAl₂O₄ forming temperature.

5. Vertically aligned ZnO nanoneedles fabricated on the ZnAl₂O₄/α-sapphire substrates

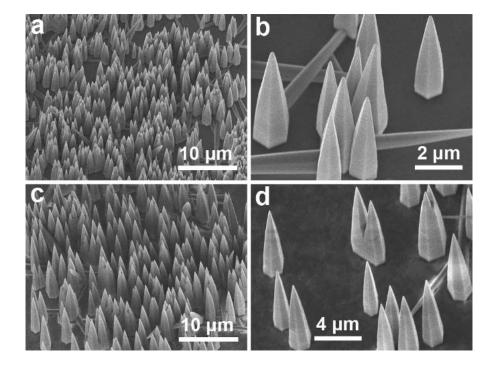


Figure S5. Based on previous reports,^{S3} we modified the morphology of the ZnO NRs to hexagonal nanoneedles through adding 1, 3-diaminopropane (DAP) into the reactive solution. The ZnO nanoneedles were fabricated on the $ZnAl_2O_4/\alpha$ -sapphire substrates which were prepared at 900 °C. $[Zn(NO_3)_2] = 20$ mM, [DAP]: (a, b) 140 mM and (c, d) 160 mM. The nanoneedles have perfect vertical alignment, but in some cases there exits coalescence at their bottom.

6. Growth behaviors of ZnO NRs on the MgAl₂O₄ epilayer/α-sapphire substrates

In order to illustrate the generality of our surface engineering protocol on realizing the epitaxial growth of ZnO NRs, we also fabricated MgAl₂O₄ epilayers on α -sapphire substrates in order to compare with the ZnAl₂O₄ counterpart on assisting the growth of ZnO NRs. MgO layers were first sputtered on the substrates using DC sputtering. After annealing in air at high temperatures (\geq 800 °C) and subsequent etching in diluted acid to remove the residual MgO, the MgAl₂O₄ layers with different roughness were obtained. We found that the magnitude of roughness tunability for MgAl₂O₄ layer is much smaller than that of ZnAl₂O₄ with the same synthesis parameters, indicating that its modulation for ZnO NR density is less effective than ZnAl₂O₄ (Figure S6). Furthermore, the hydrolysis of MgO makes the preparation of clean MgAl₂O₄ layers much more difficult. This control experiment further underscores the unique advantages of ZnAl₂O₄ as buffers layers to support the growth of high quality ZnO NRs.

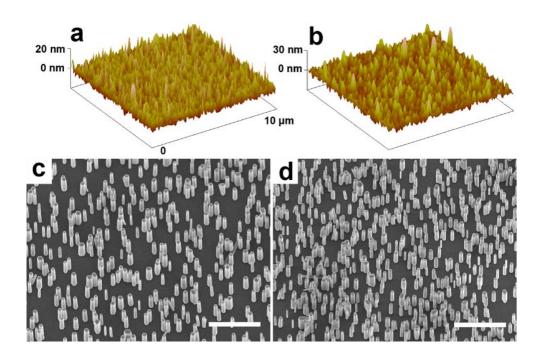


Figure S6. AFM micrographs of the MgAl₂O₄/ α -sapphire substrates produced by annealing at (a) 800 °C and (b) 900 °C, the rms roughnesses are 3.225 nm and 4.024 nm, respectively. (c) and (d) Corresponding SEM images of the ZnO NRs grown on the substrates (a) and (b) at 70 °C with [Zn(NO₃)₂]= 20 mM. All the scale bars represent 10 µm.

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