## **Supporting Information**

## **Oxidation Effect in Octahedral Hafnium Disulfide Thin Film**

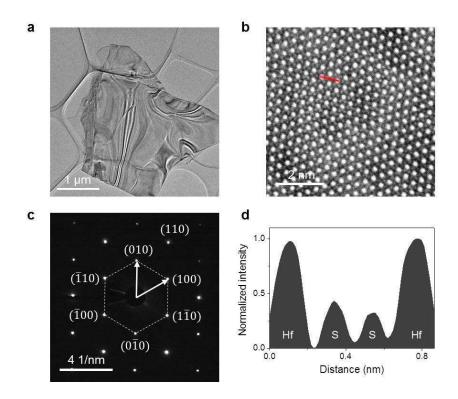
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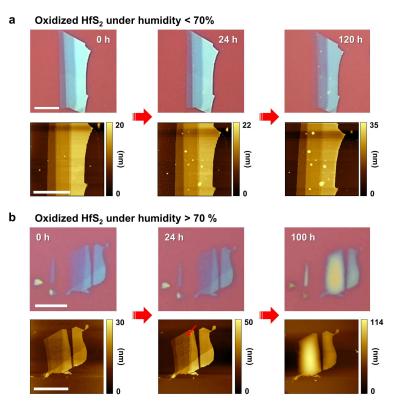
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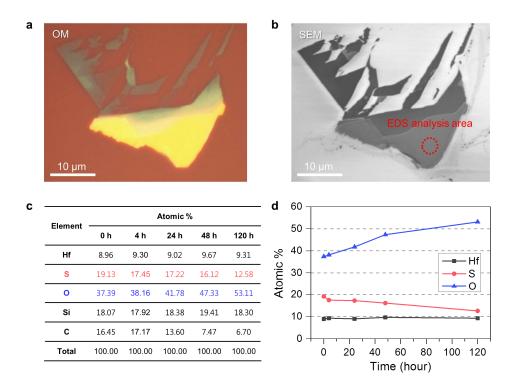
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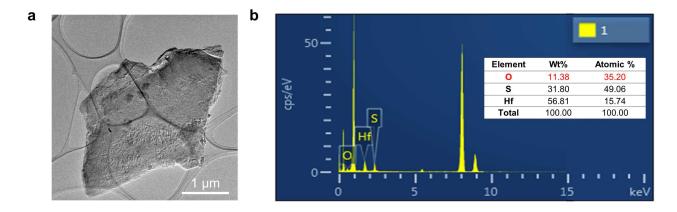
**Figure S1.** Transmission electron microscopy (TEM) measurement of  $HfS_2$ . (a) TEM image of exfoliated  $HfS_2$ . Exfoliated  $HfS_2$  was deposited onto carbon grids inside a Cu TEM grid. (b) Unfiltered scanning transmission electron microscopy (STEM) image of  $HfS_2$  showing that Hf and S atoms contrast clearly. Blurred parts are due to differences in thickness; however, there is no aggregation. (c) Selective-area electron diffraction (SAED) image of  $HfS_2$  showing a hexagonal lattice structure with labeled symmetric points in the reciprocal lattice. Interplane distances are slightly different from the data extracted from the STEM image in the main text: 0.35 (100) and 0.20 (110) nm. (d) Normalized intensity profile from the STEM image (red line). Each atom is identified through intensity difference.



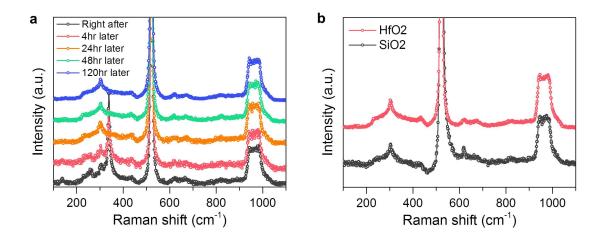
**Figure S2.** Time-dependent morphological degradation of  $HfS_2$ . (a, b) Optical (top) and topological (bottom) images of exfoliated  $HfS_2$  show degradation of each samples due to oxidation effect.  $HfS_2$  flakes were prepared by mechanical exfoliation onto SiO<sub>2</sub> (300 nm)/Si substrates. Oxidation seems to start from defect sites of the sample with faster oxidation process, so that partly more oxidized parts (protrusions) have larger thickness as shown in both images with bubble-like shape.<sup>1</sup> Finally, whole thickness increases uniformly proportional to the time. Oxidation process was conducted in the limited environment with (a) high and (b) low humidity to control the oxidation speed. Oxidation process involves oxygen and water molecules, and humidity is an important factor in oxidation of  $HfS_2$  as compared between (a) and (b). The scale bar is 10 µm.



**Figure S3.** Scanning electron microscopy (SEM) and Energy-dispersive X-ray spectroscopy (EDS) analysis for oxidized HfS2. (a) Optical and (b) SEM images of exfoliated HfS<sub>2</sub> flake. The scale bar is 10  $\mu$ m. (c) EDS result with time variation (0, 4, 24, 48, and 120 h) for the HfS<sub>2</sub> flake being exposed in air. (d) The atomic percentage of S keeps reducing under air exposure. In contrast, the atomic percentage of O keeps increasing.



**Figure S4.** TEM measurement of oxidized  $HfS_2$  ( $HfO_X$ ). (a) TEM image of oxidized  $HfS_2$ .  $HfS_2$  was exposed to air in order to measure its oxidized parts ( $HfO_X$ ). TEM image of this part seems to show a rough and aggregated surface. (b) Energy-dispersive X-ray spectroscopy (EDS) data of an oxidized part showing evidence of oxidation of  $HfS_2$  with a high-intensity oxygen component.



**Figure S5.** Raman spectra analysis for  $HfS_2$ . (a) Raman spectra of  $HfS_2$  with time variation from 100 to 1100 cm<sup>-1</sup>.  $HfS_2$  was mechanically exfoliated onto a SiO<sub>2</sub> (300 nm)/Si substrate. Raman spectra show  $HfS_2$  Raman modes with several oxidation modes. (b) Raman spectra of SiO<sub>2</sub> and  $HfO_2$ .  $HfO_2$  was deposited onto a SiO<sub>2</sub> (300 nm)/Si substrate by atomic layer deposition. Raman spectra of  $HfO_2/SiO_2/Si$  are in great accordance with those of  $SiO_2/Si$ , therefore, it is difficult to distinguish  $HfO_2$  Raman modes and its intensity change from those of  $HfS_2$  with time variation. The intrinsic peak of  $HfO_X$  are buried into  $SiO_2$  peak due to the poor crystallinity of  $HfO_X$ .<sup>2,3</sup>

References

1. Beechem, T. E.; Kowalski, B. M.; Brumbach, M. T.; McDonald, A. E.; Spataru, C. D.; Howell, S. W.; Ohta, T.; Pask, J. A.; Kalugin, N. G. Oxidation of Ultrathin Gase. *Appl. Phys. Lett.* **2015**, 107, 173103.

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3. Borowicz, P.; Taube, A.; Rzodkiewicz, W.; Latek, M.; Gieraltowska, S. Raman Spectra of High-κ Dielectric Layers Investigated with Micro-Raman Spectroscopy Comparison with Silicon Dioxide. *Sci. World J.* **2013**, 2013, 208081-208087.