

## **Supporting information**

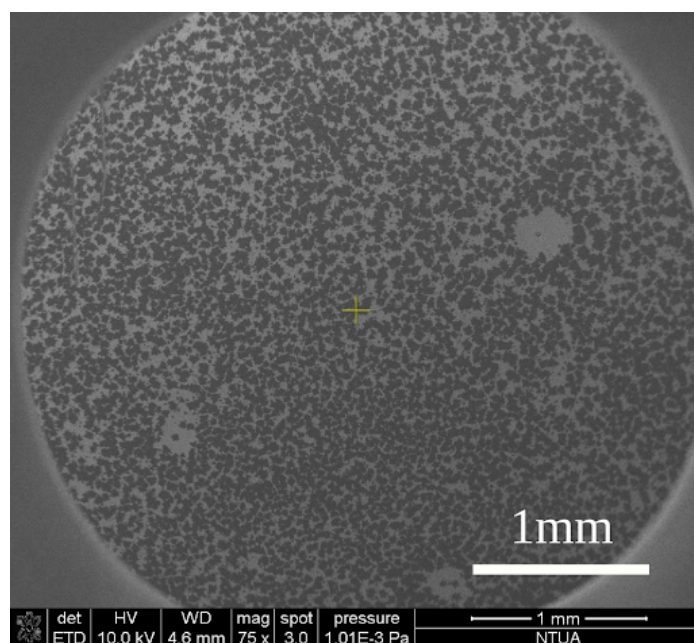
# **Demonstration of enhanced switching variability and conductance quantization properties in a $\text{SiO}_2$ conducting bridge resistive memory with embedded 2D $\text{MoS}_2$ Material**

*Stavros Kitsios<sup>\*1</sup>, Panagiotis Bousoulas<sup>1</sup>, Dimitris Spithouris<sup>1</sup>, Maria Kainourgiaki<sup>1</sup>,  
Menelaos Tsigkourakos<sup>1</sup>, Polyxeni Chatzopoulou<sup>2</sup>, George P. Dimitrakopoulos<sup>2</sup>, Philomela  
Konninou<sup>2</sup>, \*\*Dimitris Tsoukalas<sup>1</sup>*

<sup>1</sup> School of Applied Mathematical and Physical Sciences, National Technical University of Athens, Iroon Polytechniou 9, 15772, Greece

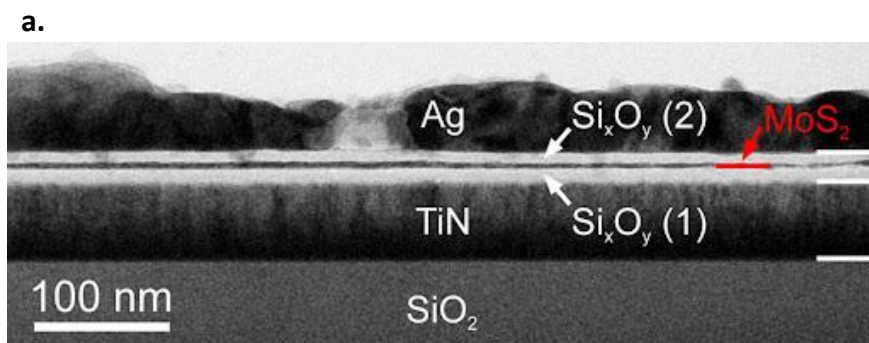
<sup>2</sup> Physics Department, Aristotle University, Thessaloniki, 54124, Greece  
E-mail: [\\*stkitsio@mail.ntua.gr](mailto:*stkitsio@mail.ntua.gr), [\\*\\*dtsouk@central.ntua.gr](mailto:**dtsouk@central.ntua.gr)

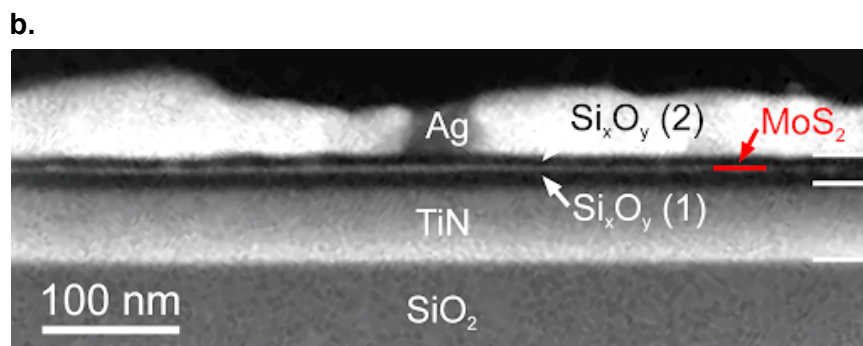
## S1. $\text{MoS}_2$ SEM images



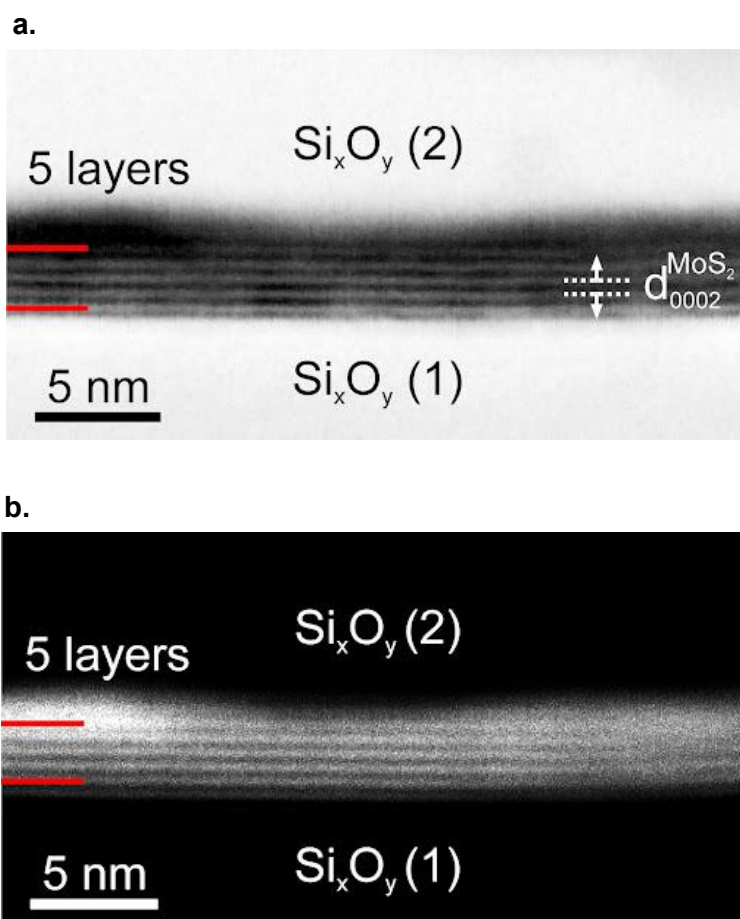
**Figure S1.** SEM image showing CVD grown  $\text{MoS}_2$  with multiple layers.

## S2. TEM Structural Analysis

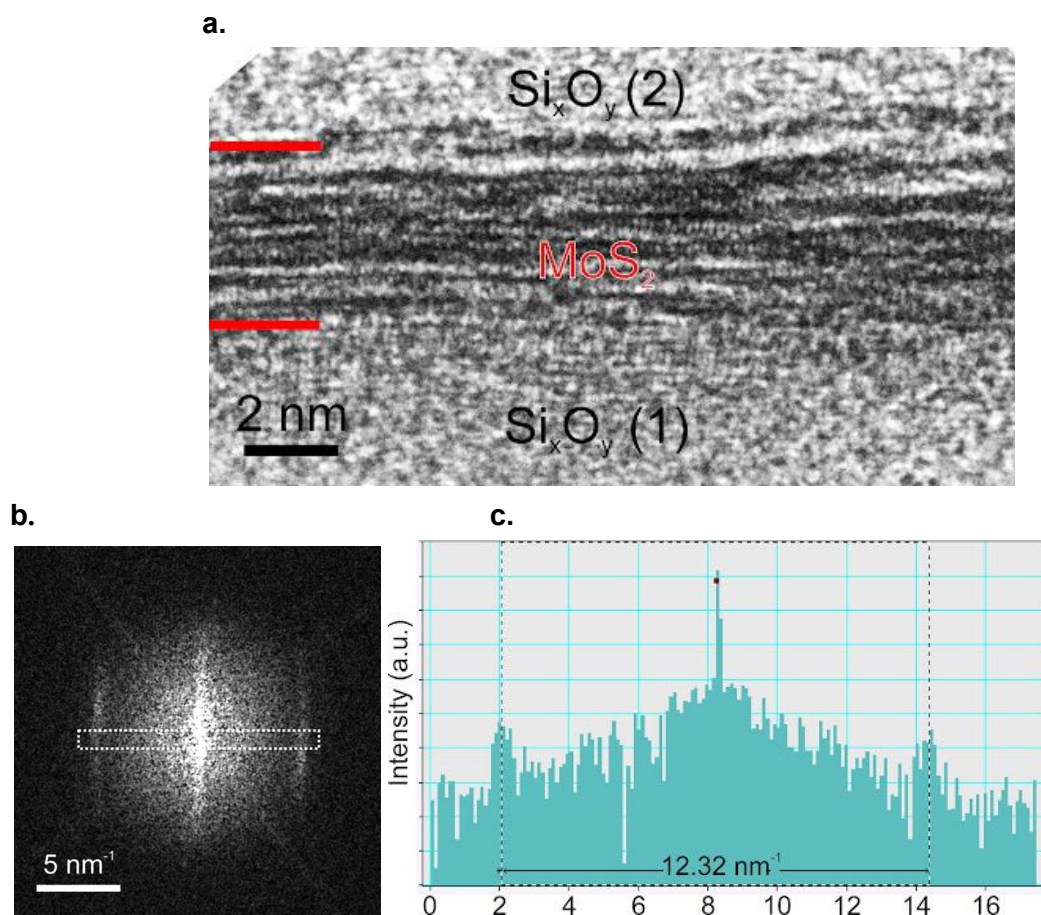




**Figure S2. a.** Cross sectional TEM (XTEM) micrograph obtained along the  $[1\bar{1}0]_{\text{Si}}$  zone axis (ZA) of the Si wafer. Few-layer of  $\text{MoS}_2$  was successfully grown on the amorphous  $\text{Si}_x\text{O}_y$  (1) layer exhibiting good surface coverage. **b.** Z-contrast high-angle annular dark field (HAADF) STEM image from around the same area. The average thickness of each alternate layer is presented in table 1.

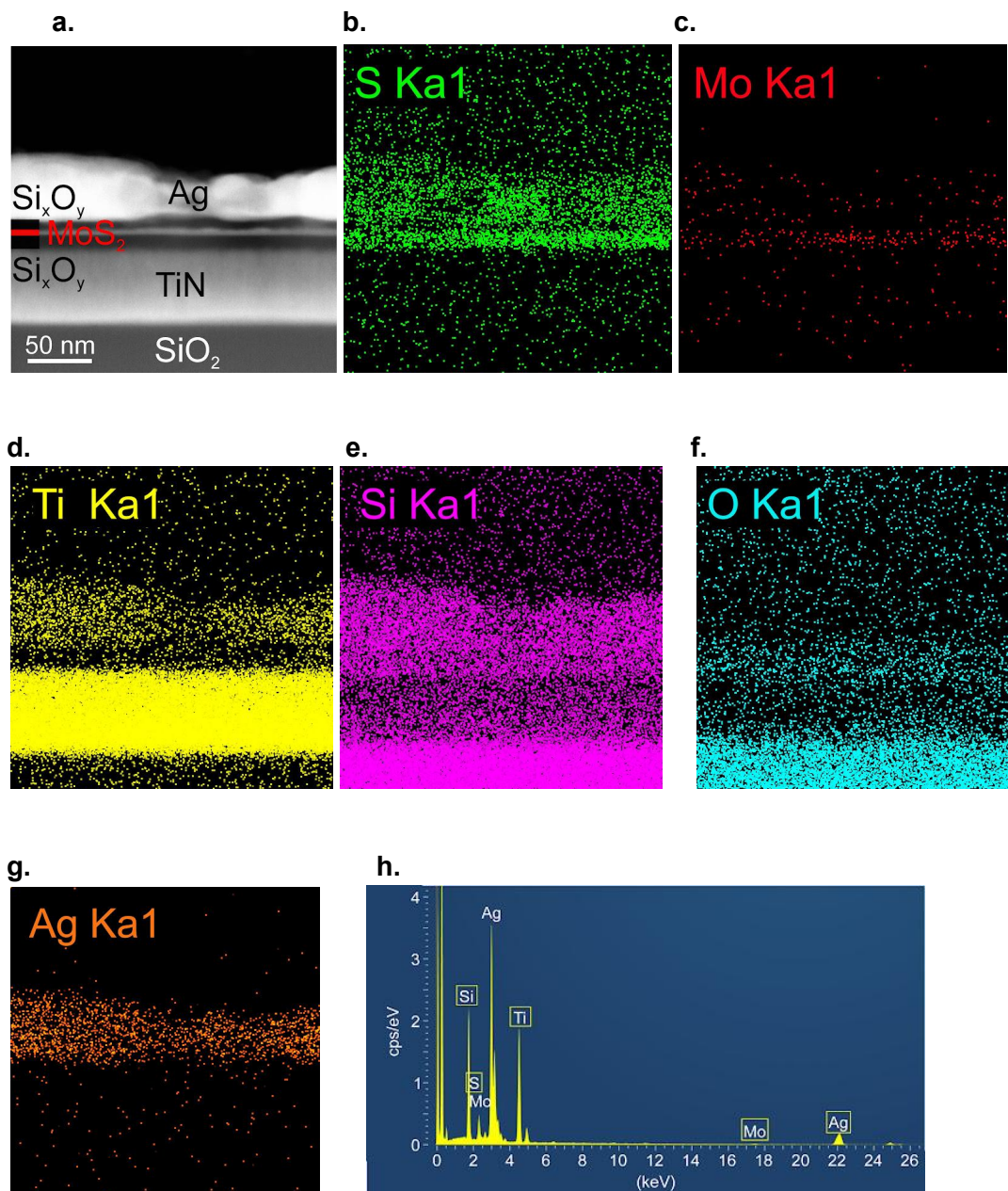


**Figure S3. a.** Annular bright field (ABF) and **b.** Z-contrast HAADF cross-sectional STEM images showing a  $\text{MoS}_2$  film consisting of five (0002) layers, with an interplanar space of  $0.65 \pm 0.5$  nm, which is close to the nominal value of hexagonal  $\text{MoS}_2$  ( $d_{0002} = 0.62$  nm).



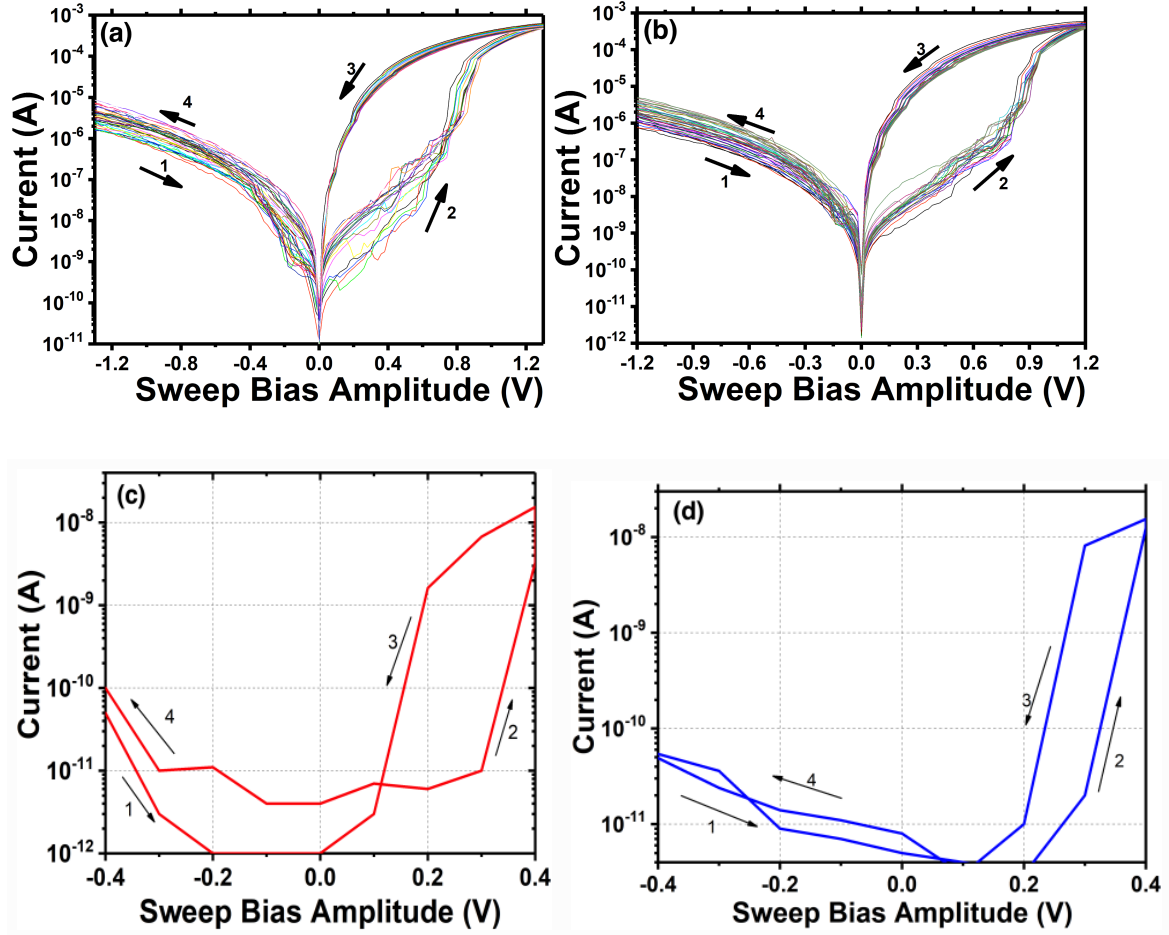
**Figure S4. a.** High magnification cross-sectional HRTEM image from the  $\text{MoS}_2$  layer. The in-plane lattice spacings are not clearly resolved, due to their low spacing and the fact that  $\text{MoS}_2$  is not well ordered, as it grown on top of an amorphous  $\text{Si}_x\text{O}_y$  layer and may exhibit tilted grains. **b.** Diffraction pattern obtained from the HRTEM image by Fourier transform and, **c.** corresponding intensity profile from the delineated region in b. The in-plane g-vector was measured at  $\sim 6.16 \text{ nm}^{-1}$ , corresponding to 0.16 nm in direct space, which is consistent with the  $d_{112\bar{0}}$  interplanar spacing of  $\text{MoS}_2$ .





**Figure S5.** **a.** STEM image from the multilayer structure, and **b-g.** Elemental maps obtained by EDX. The appearance of Si, S, and Ti inside the Ag layer is attributed to background scattering as well as to the influence of the ion milling during TEM sample preparation. **h.** Overall EDX spectrum from this area. The lines used for the elemental maps are indicated.

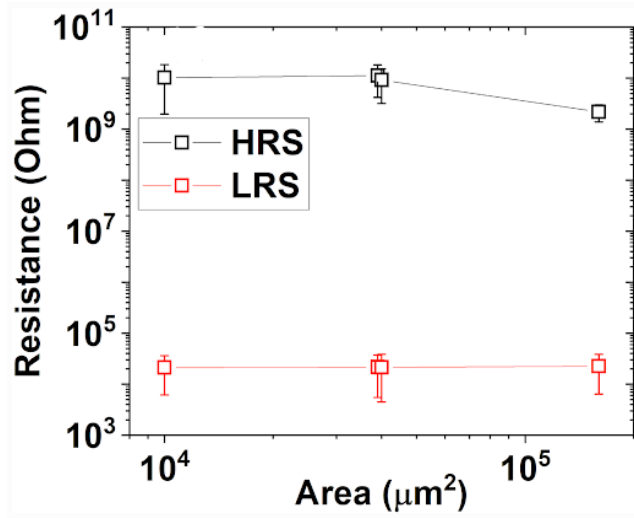
### S3. I-Vs



**Figure S6.** a) I-V cycles for the reference sample and b) for the MoS<sub>2</sub> embedded sample, c) I-V cycles for lower compliance currents for the reference sample and d) for the MoS<sub>2</sub> embedded sample indicating the threshold switching effect.

### S4. Device Area and Temperature Dependence

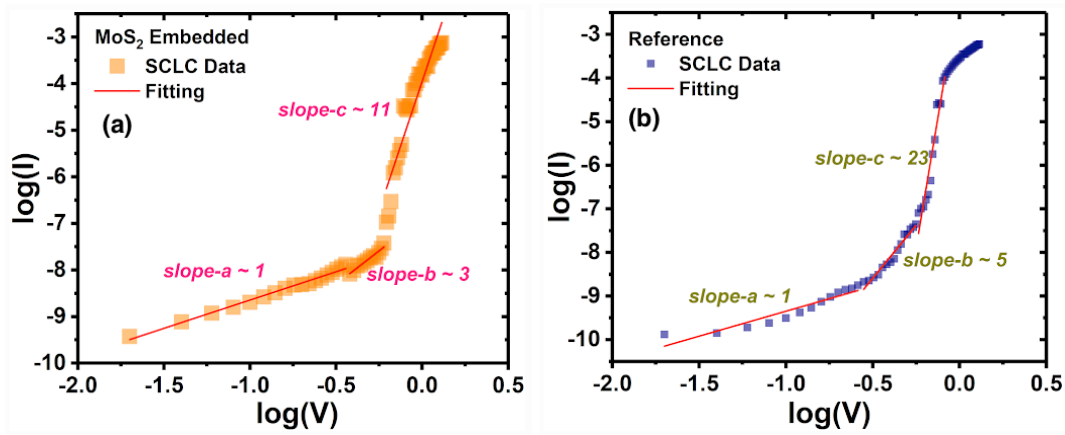
Insights regarding the potential existence of CFs within our device's active core are presented below. More specifically, the total independence of the LRS distribution from the total device area (Figure S7a) provides strong pieces of evidence towards the formation of percolating CFs.



**Figure S7.** Distribution of both HRS and LRS as a function of the total device area.

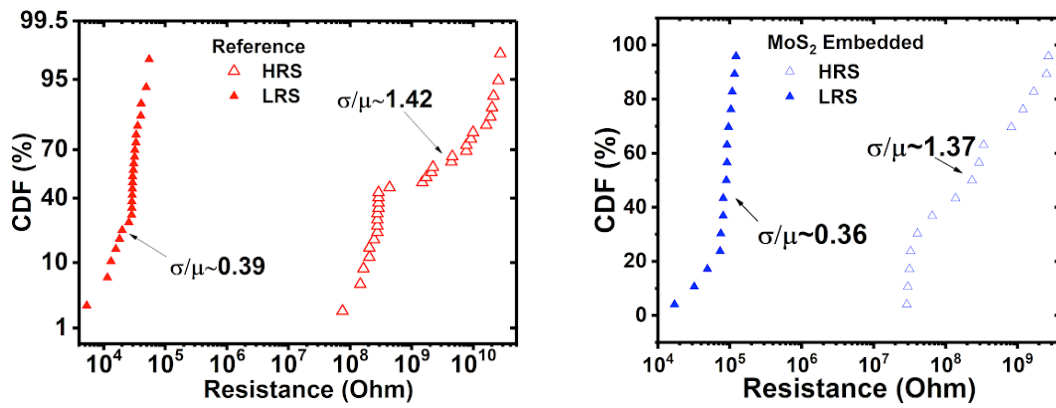
### S5. Conduction Mechanism

A further examination of the conduction mechanism was done in order to gain insight of how the thin films of CVD grown  $MoS_2$  affect the behavior of the memory device. For this reason, distributions of current and voltage during the SET process were plotted in logarithmic scale. This is an attribute of the Space-Charge-Limited-Current (SCLC) mechanism whose function is based on nonlinear features of the IV characteristics. In our case, figures and represent the forward part of the IVs, for reference and  $MoS_2$  embedded stacks respectively. We observed three different segments of the distribution, two linear that are represented by slope-a and slope-c and the nonlinear increase region (slope-b). The first region refers to  $V < 0.5V$ , the nonlinear region expands from 0.5V to 1V with the third occur at  $V > 1V$ . Theory states that the first region with slope-a is represented by the proportionality of current and voltage ( $I \propto V$ ) and reflecting ohmic behavior while the third region with slope-c implements the Child's law ( $I \propto V^2$ ).<sup>1</sup>



**Figure S8.** I-Vs representing SCLC mechanism for MoS<sub>2</sub> embedded (a) and reference samples (b) respectively.

#### S6. CDF device-to-device measurements



**Figure S9.** Device-to-device cumulative distribution functions.



### S7. Device-to-device endurance and retention measurements for MoS<sub>2</sub> embedded samples

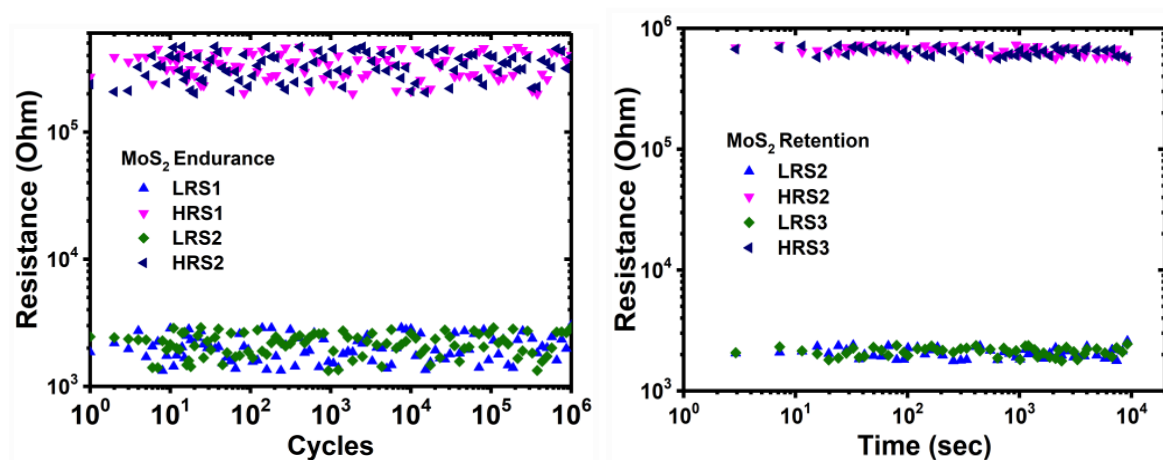


Figure S10. Device-to-device endurance and retention for MoS<sub>2</sub> embedded samples.

### S8. SiO<sub>2</sub> various thicknesses and CBRAM performance

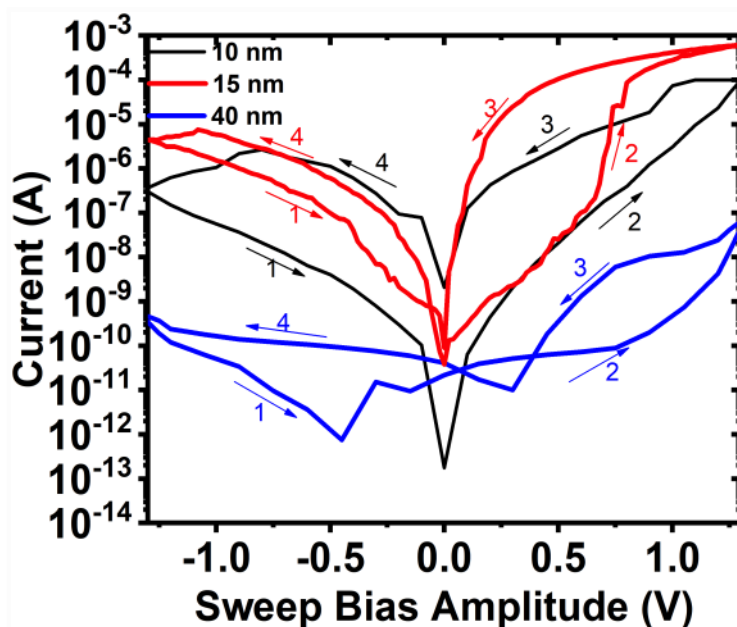
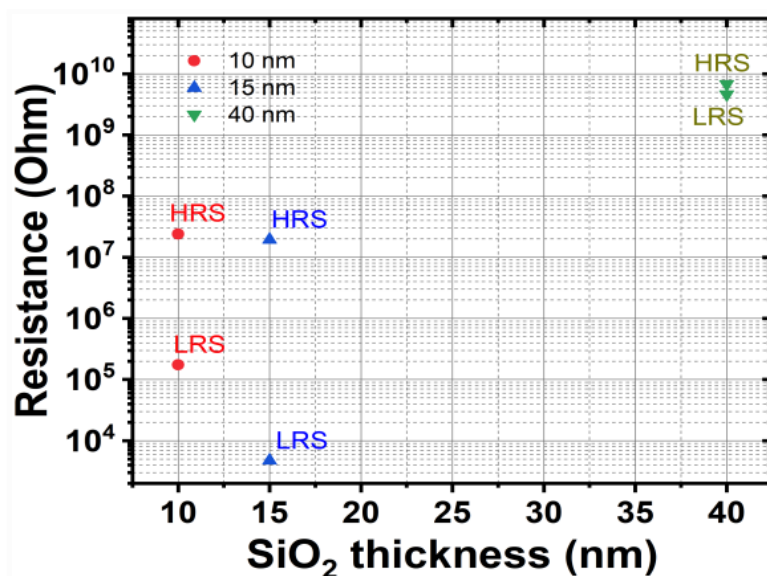


Figure S11. I-V characteristics for various SiO<sub>2</sub> thicknesses (10 nm, 15 nm and 40 nm).



**Figure S12.** HRS and LRS for the various SiO<sub>2</sub> thicknesses.

## References

1. Voronkovskii VA, Aliev VS, Gerasimovaet AK, Islamov DR. Conduction mechanisms of TaN/HfOx/Ni memristors. *Mater. Res. Express*. 2019, **6**, 076411.