Supporting Information

Heteroatom Doping-Induced Defected Co₃O₄ Electrode for High-

Performance Lithium Oxygen Battery

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Experimental section

Materials synthesis

Synthesis of Co₃O₄

The typical synthetic procedure of Co₃O₄ was modified on the basis of the previous report.¹⁻² Firstly, 0.036 mol cobalt nitrate hexahydrate (Co(NO₃)₂·6H₂O) and 0.009 mol sodium hydroxide (NaOH) were dissolved in 30 ml and 5 ml of deionized water, respectively. After stirring for 0.5 h, NaOH solution was added into Co(NO₃)₂ solution dropwise and the mixed solution went on stirring for another 0.5 h, following by transferring into a 50 mL Teflon-lined autoclave. The hydrothermal process was conducted at 180 °C for 6h. The obtained products were cleaned with ethanol and deionized water for several times, subsequently dried in a vacuum oven for 12 h at 80 °C.

Synthesis of S-Co₃O₄ and P-Co₃O₄

 $P\text{-}Co_3O_4$ was fabricated through low-temperature phosphorization method, using sodium hypophosphite (NaH₂PO₂·H₂O) as phosphorus source. Concretely, the obtained Co₃O₄ product (50 mg) and NaH₂PO₂·H₂O (80 mg) were placed at two separated quartz boat with NaH₂PO₂·H₂O at the upstream side in a tube furnace. The samples were heated at 300 °C for 2 h with a ramp of 2 °C/min under slow argon gas flow, and cooled down naturally to room temperature. To prepare S-Co₃O₄, S powder (60 mg) was used as S source via the same heating procedure. In order to avoid the influence of annealing, pure Co₃O₄ sample was also heated at the same way without any NaH₂PO₂·H₂O and S powder.

Materials Characterization

The morphology of samples and discharge product were identified by SU-70 field emission scanning electron microscope (FE-SEM-70) and high resolution transmission electron microscopy (HRTEM, JEM-2100 at an acceleration voltage of 200 kV) equipped with energy dispersive spectrometry (EDS). The sample for HRTEM was cut on Leica Ultracut UCT ultramicrotome (UC7) to reduce the thickness. The phase structure and chemical components were characterized by X-ray diffraction (XRD, Rigaku D/Max-KA diffractometer using Cu Kα radiation (λ=1.5406 Å)), X-ray photoelectron spectroscopy (XPS, ESCALAB 250 spectra with 150 W Al Kα probe beam), Raman spectroscopy (JY HR800), ICP Optical emission spectrometer Varian 720-ES (ICP-ES) and Fourier transform infrared spectroscopy (FTIR, Bruker Vector 22 system). The electron spin resonance (ESR) spectra were recorded using a Bruker EMX spectrometer (X band) at room temperature.

Electrochemical Measurements

The electrochemical performances of Co₃O₄, S-Co₃O₄ and P-Co₃O₄ were evaluated using 2032 cointype cells with gas holes. The electrode was prepared by dissolving each of the as-prepared sample (50 wt%), carbon black (40 wt%) and polyvinylidene fluoride (PVDF, 10 wt%) in N-Methyl pyrrolidone (NMP) and dip-coating the obtained slurry on carbon paper. After drying at 80 °C for 24 h, the as-prepared electrode was assembled with Li metal anode, glass fiber separator (Whatman, GF/D) and organic electrolyte (1 M LiTFSI/TEGDME) in an argon-filled glove box (H₂O< 0.5 ppm, O₂< 0.5 ppm). The mass loading of cathode is about 1 mg cm⁻². All of the electrochemical measurements were carried out under a 1.0-bar dry oxygen atmosphere (> 99.999%) at room temperature. Galvanostatic discharge profiles were measured on LAND CT2001A battery system (Wuhan, China) within the voltage window of 2.0-4.5 V. Cyclic voltammetry (CV, within the voltage window of 2.0-4.5 V at 0.1 mV s⁻¹ speed) and electrochemical impedance spectroscopy (EIS, in a frequency range of 0.01–10⁵ Hz.) analysis were performed on an electrochemical workstation (PARSTAT2273).

Computational Details

The Vo formation energy and LiO₂ adsorption energy on (100) plane of each samples were investigated using the Vienna ab initio simulation packages (VASP) with the projector augmented wave (PAW) method.³⁻⁴ The exchange-correlation energy was described by the generalized gradient approximation Perdew-Burke-Ernzerhof (GGA-PBE). The simulated surfaces were separated by a vacuum slab of 15 Å in c-axis to avoid the interactions between the periodically repeated images. The plane wave kinetic energy cutoff and the convergence criterion were set to 420 eV and 1×10^{-6} eV, respectively. The Brillouin zone was sampled by a $3 \times 3 \times 3$ k-point mesh for crystals and slabs calculations. The atomic geometries were relaxed until Hellmann–Feynman forces on each atom are reduced to less than 0.05 eV Å⁻¹ during structure optimization. The adsorption energy (ΔE_{ads}) was denoted as: $\Delta E_{ads} = E_{sub+LiO_2} - E_{sub} - E_{LiO_2}$, where $E_{sub+LiO_2}$ stands for the sum energy of the optimized substrate/LiO₂ system, E_{sub} and E_{LiO_2} are on behalf of the total energies of the bare substrate and isolated LiO₂, respectively.

Supporting Figures and Tables

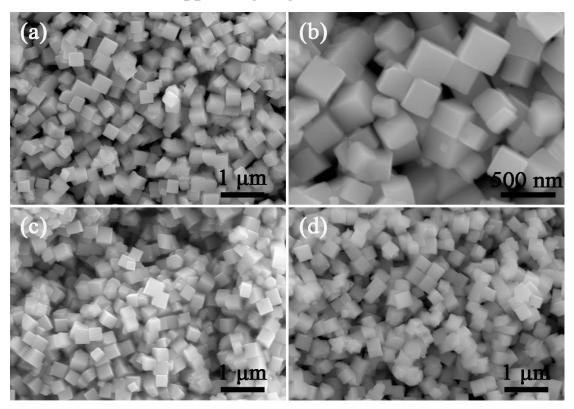


Figure S1. SEM images of (a, b) Co_3O_4 (c) S- Co_3O_4 and (d) P- Co_3O_4 .

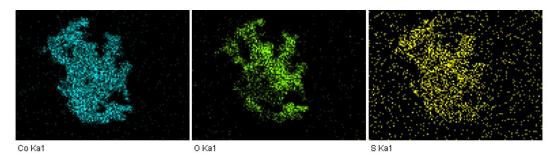


Figure S2. EDS mapping of S-Co₃O₄.

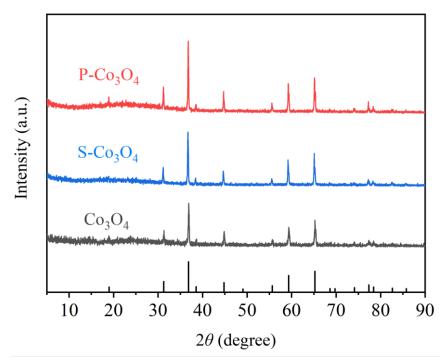


Figure S3. XRD patterns of Co₃O₄, S-Co₃O₄ and P-Co₃O₄.

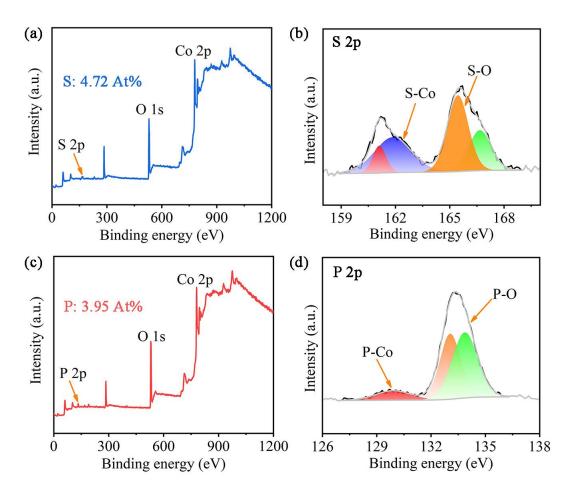


Figure S4. (a) XPS survey spectrum and (b) High-resolution XPS S 2p spectra of S-Co₃O₄. (c) XPS survey spectrum and (d) High-resolution XPS P 2p spectra of P-Co₃O₄.

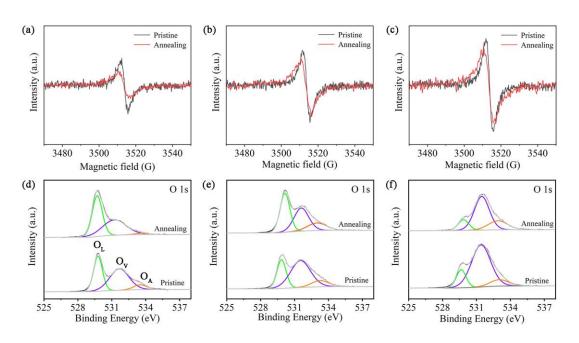


Figure S5. The characterization of stability of Vo for three samples. EPR spectra of (a) Co₃O₄, (b) S-Co₃O₄ and (c) P-Co₃O₄ before and after annealing in air. High-resolution XPS O 1s spectra of (a) Co₃O₄, (b) S-Co₃O₄ and (c) P-Co₃O₄ before and after annealing in air.

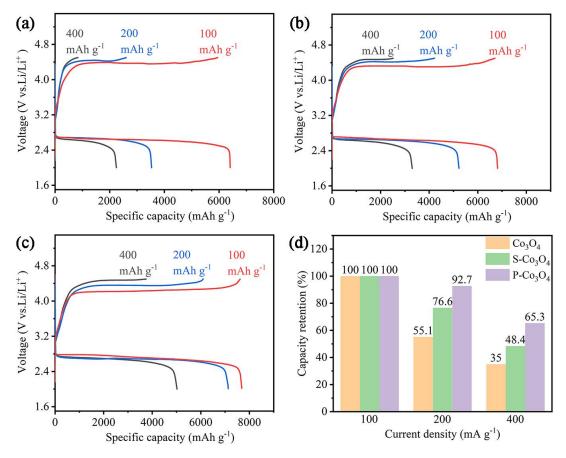


Figure S6. Discharge-charge curves at different current densities. (a) Co₃O₄. (b) S-Co₃O₄. (c) P-Co₃O₄. (d) The capacity retention of the three kinds of electrodes.

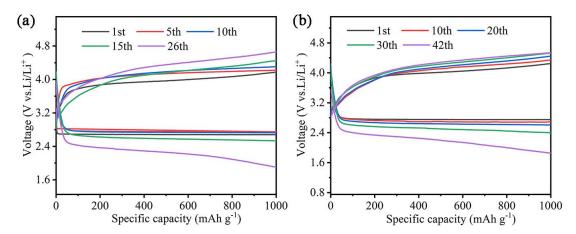


Figure S7. Discharge-charge profiles with different cycles at 100 mA g⁻¹. (a) Co₃O₄. (b) S-Co₃O₄.



Figure S8. Photo image of the green LED powered by P-Co₃O₄-based Li-O₂ battery.

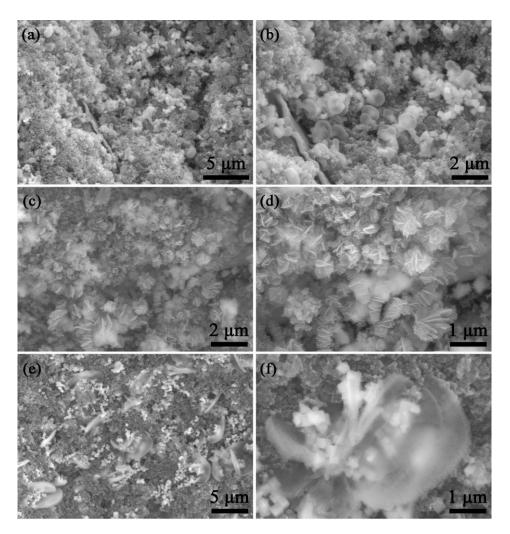


Figure S9. SEM images of three electrodes after fully discharged with different magnitude. (a, b) Co₃O₄. (c, d) S-Co₃O₄. (e, f) P-Co₃O₄.

Comments: The order of Figure S9 c/d and e/f is reverse. Therefore, we exchanged the the order of Figure S9 c/d and e/f. We confirm that the deletion will not do harm to the integrality and veracity of this work.

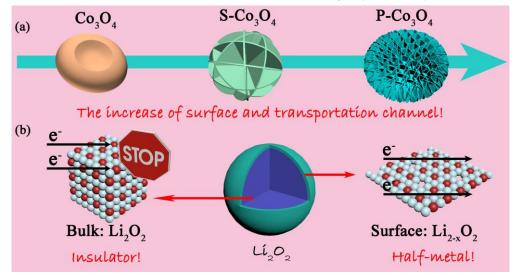


Figure S10. (a) The morphology changes of Li₂O₂ for Co₃O₄, S-Co₃O₄ and P-Co₃O₄. (b) Schematic illustration of electronic conductivity for Li₂O₂ surface and inside bulk Li₂O₂.

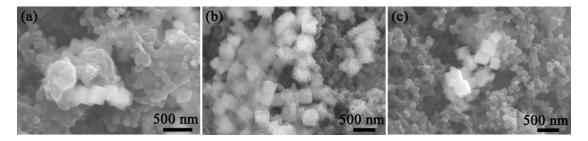


Figure S11. Ex-situ SEM images of three electrodes after fully recharged. (a) Co₃O₄. (b) S-Co₃O₄ and (c) P-Co₃O₄.

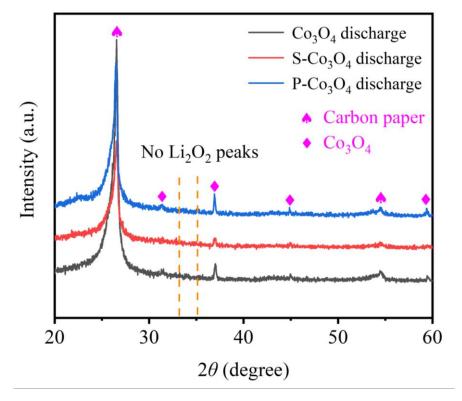


Figure S12. XRD patterns of three electrodes after first full recharge.



Figure S13. The corresponding equivalent circuit for EIS analysis.

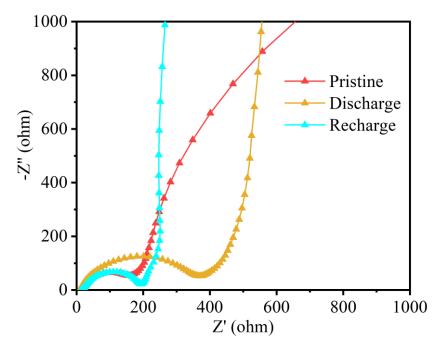


Figure S14. Nyquist plots of P-Co₃O₄ cathode at pristine and full discharge/recharge stage.

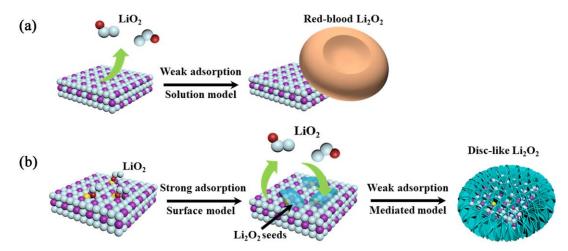


Figure S15. Schematic illustrations of the working mechanism for the (a) Co₃O₄ and (b) P-Co₃O₄ electrodes.

Table S1. Elemental contents of S-Co₃O₄ and P-Co₃O₄.

S or P (At%)	ICP-MS	EDS	XPS	
S-Co ₃ O ₄	1.21	0.54	4.72	
P-Co ₃ O ₄	1.16	0.64	3.95	

The atomic percent contents measured by ICP-ES, EDS and XPS are quite different due to their intrinsic characteristics. The elemental contents given by ICP-ES reflect the true value of entire sample. However, XPS can only measure the depth of a few nanometers of sample surface. The testing depth EDS is larger than XPS but the value measured by EDS is not accurate. In view of heteroatom doping mainly on the surface, XPS results are closer to reality of doping than ICP-MS and EDS results.

Table S2. Comparison of electrochemical performance between some reported Co₃O₄-based cathodes for Li-O₂ batteries.

Catalysts	Over-potential (ORR/OER)	Discharge capacity (mAh g ⁻¹)	Cycle number/ cutoff capacity	Reference
Co ₃ O ₄ -(111)	~1.5	2463	20/1000	5
Co ₃ O ₄ NS	-	11882	80/500	6
Co ₃ O ₄ NFs/GNF	1.26	10500	80/1000	7
Co ₃ O ₄ -H	~1.7	5220	110/500	8
Co ₃ O ₄ -(112)	~1.3	9144	50/500	9
Co ₃ O ₄ @Co ₃ O ₄ /Ag	~1.5	12000	80/1000	10
P-Co ₃ O ₄	1.2	7690	90/1000	This

References

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