

## **Supporting Information**

### **Insulator-to-Metal Transition of Cr<sub>2</sub>O<sub>3</sub> Thin Films via Isovalent Ru<sup>3+</sup>**

#### **Substitution**

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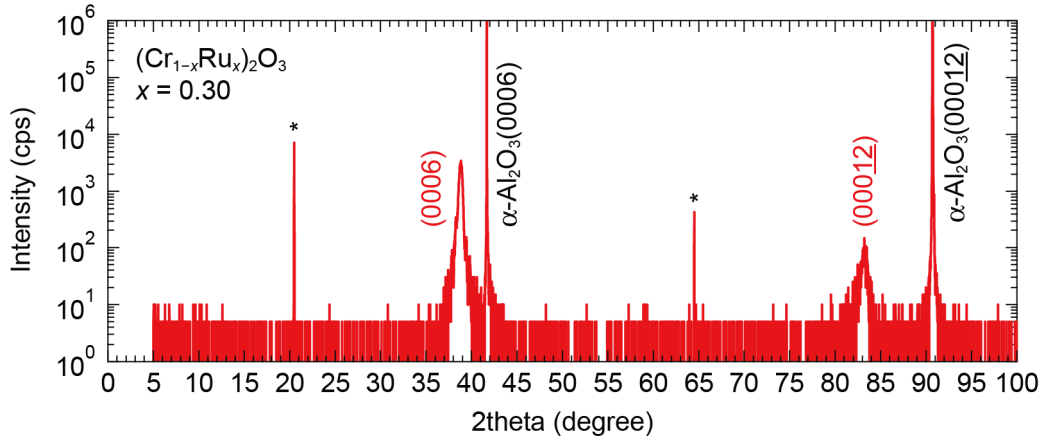
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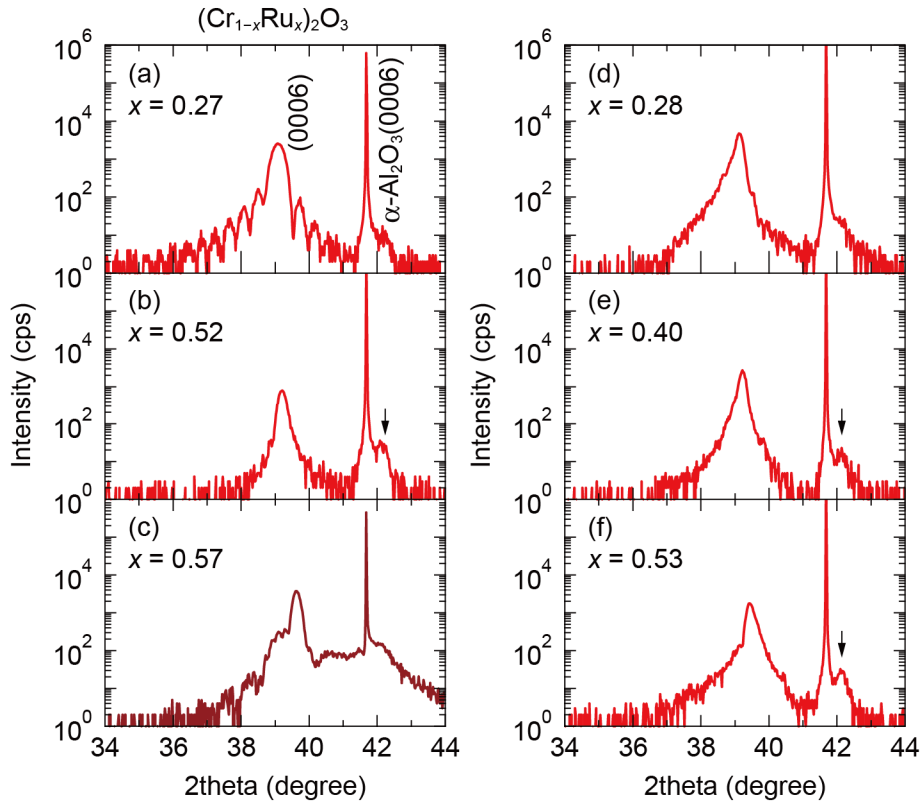
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**Table S1.** Growth conditions,  $x$  values, and  $t$  values for some of the  $(\text{Cr}_{1-x}\text{Ru}_x)_2\text{O}_3$  films used in this study. The deposition duration was not constant. XRD: X-ray diffraction, E: electrical measurements, M: magnetization measurements, TEM: transmission electron microscopy, PES: photoemission spectroscopy, and XAS: X-ray absorption spectroscopy.

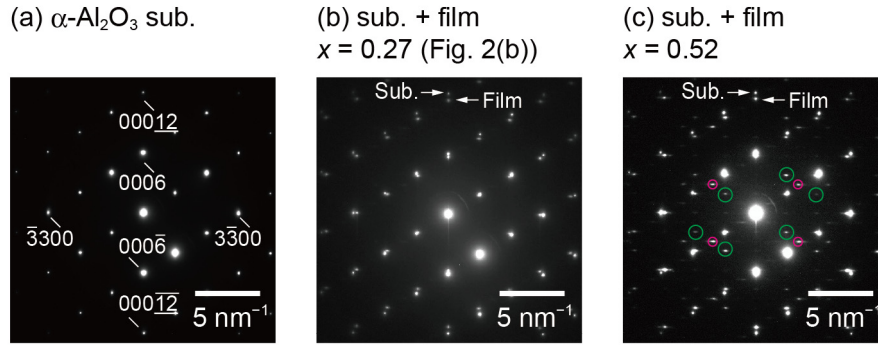
Sample ID	Nominal Ru / Cr ratio in targets	$T_g$ ( $^{\circ}\text{C}$ )	Ru content $x$ by SEM-EDX	Thickness $t$ (nm)	Experiment
#1	0	600	0	18	XRD
#2	1.0	600	0.30	23	XRD
#3	1.0	700	0.27	25	XRD, TEM
#4	1.7	670	0.52	43	XRD, TEM
#5	2.0	600	0.57	70	XRD
#6	1.0	800	0.16	16	E
#7	1.0	800	0.21	90	E, M
#8	1.0	700	0.28	60	XRD, E, XAS
#9	1.7	750	0.30	70	E, M
#10	1.7	750	0.34	85	E, M
#11	1.7	735	0.40	80	XRD, E, XAS
#12	1.7	735	0.40	95	E, M
#13	1.7	675	0.47	85	E, M
#14	2.0	720	0.53	70	XRD, E, PES, XAS
#15	0	600	0	117	M



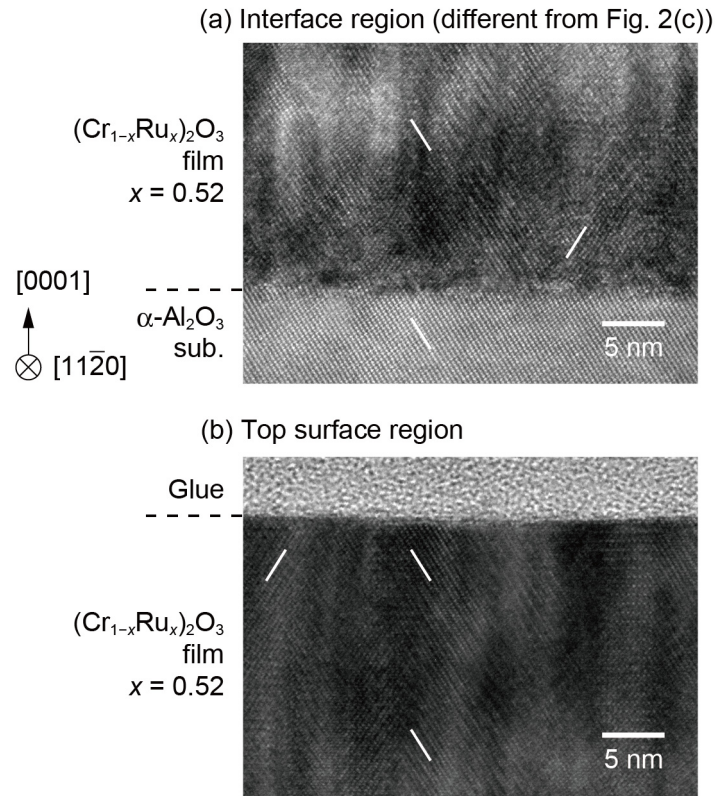
**Figure S1.** Wide-range out-of-plane XRD pattern for a  $(\text{Cr}_{0.70}\text{Ru}_{0.30})_2\text{O}_3$  film (sample #2). The asterisks indicate forbidden reflections of  $\alpha\text{-Al}_2\text{O}_3$ .



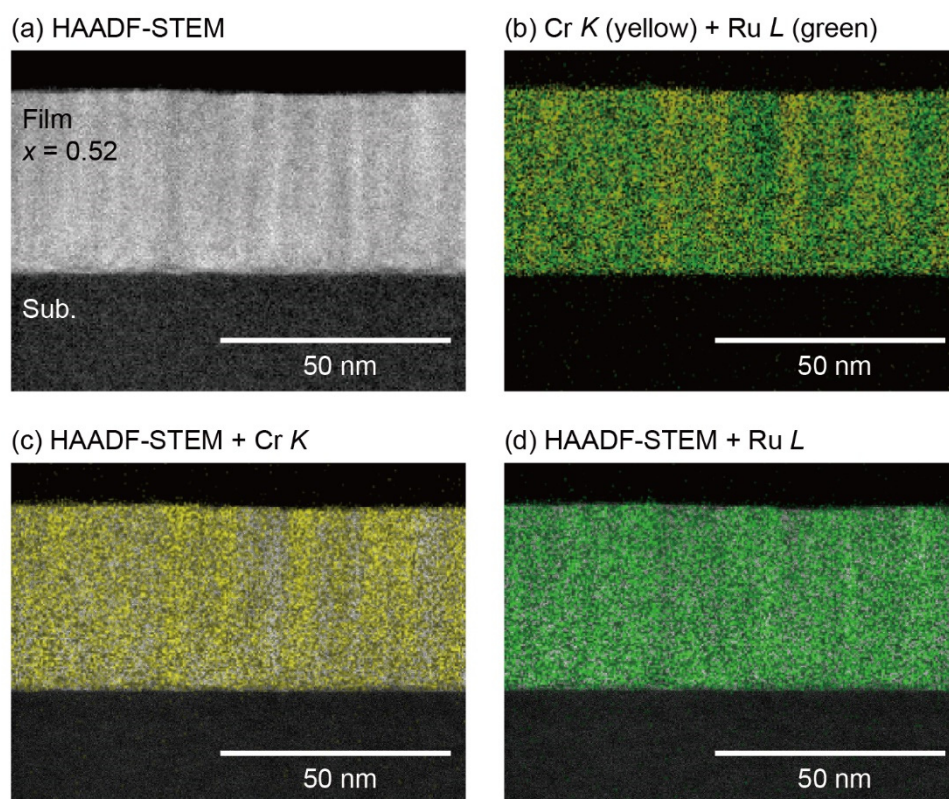
**Figure S2.** XRD patterns for the  $(\text{Cr}_{1-x}\text{Ru}_x)_2\text{O}_3$  films used for (a), (b) TEM analysis in Fig. 2, (c) a solubility limit check, and (d)–(f) spectroscopic measurements shown in Figs. 4 and 5. These films correspond to samples #3, 4, 5, 8, 11, and 14, respectively. With the exception of (c), the films exhibited sharp (0006) diffraction peaks. The arrows indicate an unidentified peak at approximately  $2\theta = 42.1^\circ$ . The asymmetry of diffraction peak seen in (c), (d), and (f) may reflect the structural inhomogeneity in the films.



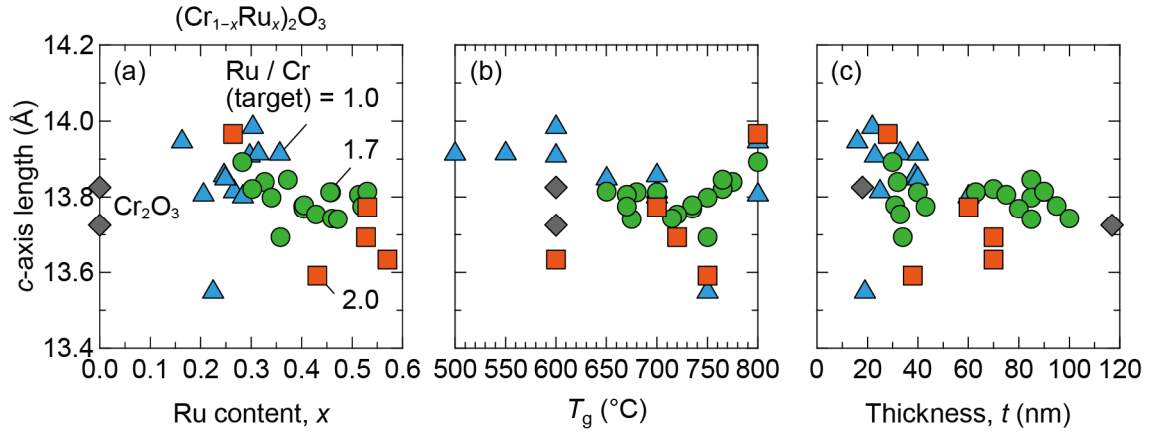
**Figure S3.** Selected-area diffraction patterns of (a) the  $\alpha$ - $\text{Al}_2\text{O}_3$  substrate region, (b) the film and substrate regions of a  $(\text{Cr}_{0.73}\text{Ru}_{0.27})_2\text{O}_3$  film (sample #3), and (c) the film and substrate regions of a  $(\text{Cr}_{0.48}\text{Ru}_{0.52})_2\text{O}_3$  film (sample #4). The data were acquired over an area of  $\sim$  a few hundreds of nm in diameter. As indicated by large green circles in (c), diffraction spots corresponding to the inverted domains appear for  $x = 0.52$ , consistent with the real-space TEM image in Fig. 2(c). Unidentified diffraction spots indicated by small pink circles might come from precipitates that are often generated in the pulsed-laser deposition process.



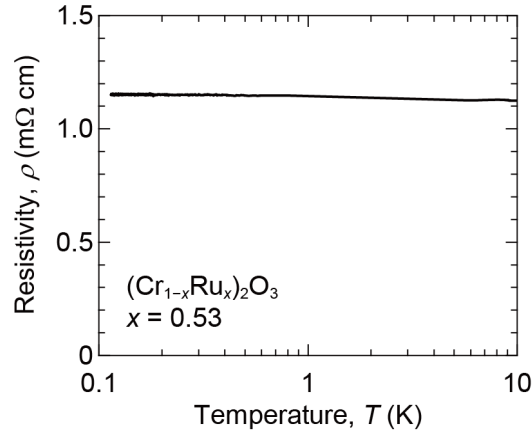
**Figure S4.** Bright-field TEM images of a  $(\text{Cr}_{0.48}\text{Ru}_{0.52})_2\text{O}_3$  film (sample #4) viewed along the  $[11\bar{2}0]$  direction: (a) the interface region (different from that shown in Fig. 2(c)) and (b) the top surface region. The  $(1\bar{1}02)$  plane characteristic of the corundum-type structure (indicated by slanting white lines) is commonly observed in all regions.



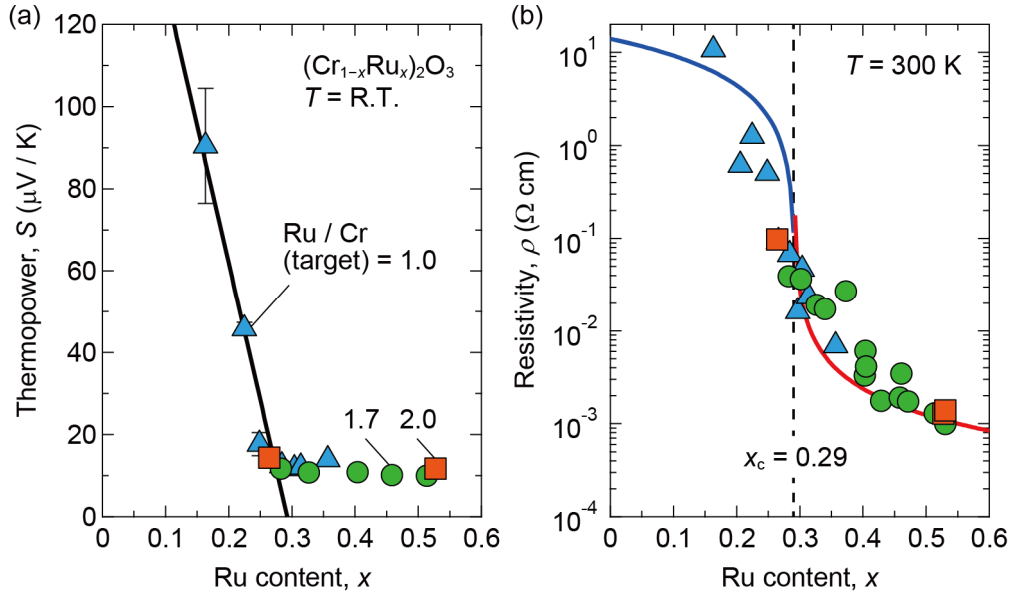
**Figure S5.** (a) High-angle annular dark-field scanning TEM (HAADF-STEM) image taken for a  $(\text{Cr}_{0.48}\text{Ru}_{0.52})_2\text{O}_3$  film (sample #4). Overlaid images of (b) Cr  $K$  (yellow) and Ru  $L$  (green) mapping results (count mode), (c) HAADF-STEM and Cr  $K$ , and (d) HAADF-STEM and Ru  $L$ . It is clear that the Ru-rich (green) region is not specific to the interface or the top surface of the film.



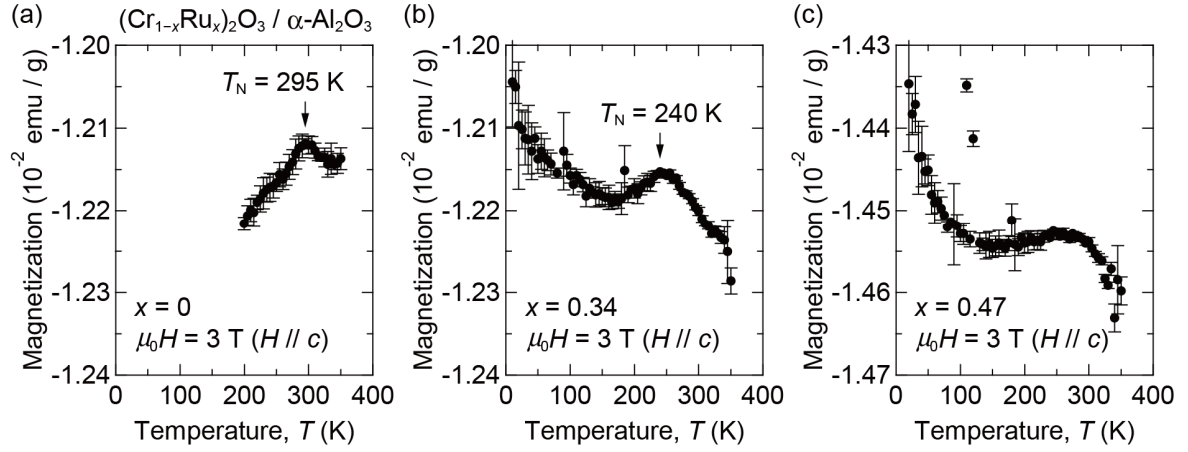
**Figure S6.** (a)  $x$  dependence of  $c$ -axis length ( $c_{\text{XRD}}$ ) for  $(\text{Cr}_{1-x}\text{Ru}_x)_2\text{O}_3$  films fabricated using targets with nominal Ru / Cr ratios of 1.0, 1.7, and 2.0. The results of non-doped  $\text{Cr}_2\text{O}_3$  films (samples #1 and 15) are also included for comparison. The  $c$ -axis length was calculated from the  $2\theta$  values of (0006) XRD peaks by assuming that the films were crystallized in the corundum structure. (b)  $T_g$  dependence and (c)  $t$  dependence of  $c$ -axis length for the same sample set. There are no clear systematic changes in  $c$ -axis length as  $x$ ,  $T_g$ , and  $t$  vary.



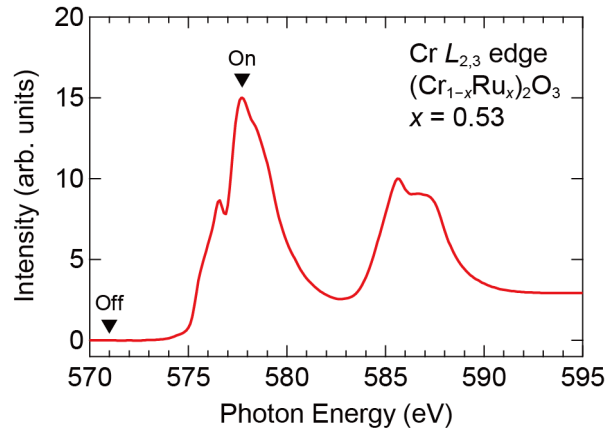
**Figure S7.**  $\rho$  versus  $T$  characteristics of a metallic  $(\text{Cr}_{0.47}\text{Ru}_{0.53})_2\text{O}_3$  film (sample #14) as evaluated by adiabatic demagnetization refrigerator (ADR) measurements. The XRD pattern is shown in Fig. S2(f).



**Figure S8.** Fitting results for (a) thermopower  $S$  and (b) resistivity  $\rho$  based on the percolation model<sup>37,38</sup> assuming conducting  $\text{RuO}_2$  and non-conducting  $\text{Cr}_2\text{O}_3$  columnar domains grown perpendicular to the  $\alpha\text{-Al}_2\text{O}_3$  substrate. The data are the same as those shown in Fig. 3. Calculating a linear fit for the  $S$  data in the low- $x$  region (the bold black line in (a)) using Eq. 7 in Ref. 38, we first determined the percolation threshold  $x_c$  to be 0.29. Inserting the  $x_c$  value into Eq. 5 in Ref. 38, we analytically obtained a fitting curve for  $\rho$  below  $x_c$ , with  $\rho(x=0)$  of  $14 \Omega \text{ cm}$ , as shown by the bold blue curve in (b). However, this curve does not reproduce the trend of variation in  $\rho$ . Furthermore, the fitting parameter of  $\rho(x=0) = 14 \Omega \text{ cm}$  is too low for highly resistive  $\text{Cr}_2\text{O}_3$  films ( $x=0$ ) with  $\rho \gg 10^3 \Omega \text{ cm}$  (as checked by the two-probe method). The bold red curve shows a fitting curve for the  $\rho$  above  $x_c$ , which also deviates from the  $\rho$  data, in particular for  $0.30 < x < 0.40$ . The actual  $\rho$  values seem to decrease more gradually. Judging from these discrepancies, we conclude that the simple percolation model is not adequate for explaining the  $x$  dependence of  $\rho$  in  $(\text{Cr}_{1-x}\text{Ru}_x)_2\text{O}_3$  films.



**Figure S9.** Magnetization versus temperature  $T$  curves measured for (a)  $\text{Cr}_2\text{O}_3$  (sample #15), (b)  $(\text{Cr}_{0.66}\text{Ru}_{0.34})_2\text{O}_3$  (sample #10), and (c)  $(\text{Cr}_{0.53}\text{Ru}_{0.47})_2\text{O}_3$  (sample #13) films in an out-of-plane magnetic field of  $\mu_0 H = 3$  T ( $\mu_0$  is the vacuum permeability and  $H$  is the magnetic field strength). Error bars indicate standard deviations of the data. The magnetization, including a diamagnetic contribution from the substrate, was normalized using the total sample weight (i.e., the total weight of the film and the 0.348 mm-thick  $\alpha\text{-Al}_2\text{O}_3$  substrate). The total sample weights of samples #15, 10, and 13 were 0.07202 g, 0.06457 g, and 0.06055 g, respectively. The upturns at low  $T$  in (b) and (c) were also seen for a bare  $\alpha\text{-Al}_2\text{O}_3$  substrate that experienced a similar heating process in the PLD chamber. Black arrows indicate local magnetization maximums used for the determination of  $T_N$ .



**Figure S10.** Cr  $L_{2,3}$  XAS data measured for a metallic  $(\text{Cr}_{0.47}\text{Ru}_{0.53})_2\text{O}_3$  film (sample #14). The spectral features are in excellent agreement with those of  $\text{Cr}^{3+}$ .<sup>41</sup> On-resonance and off-resonance photon energies for Cr  $2p\text{-}3d$  resonant PES (Fig. 4(b)) are indicated by black triangles.