

Supporting information for

# Reduced $\text{Na}_{2+x}\text{Ti}_4\text{O}_9/\text{C}$ composite: a durable anode for sodium-ion batteries

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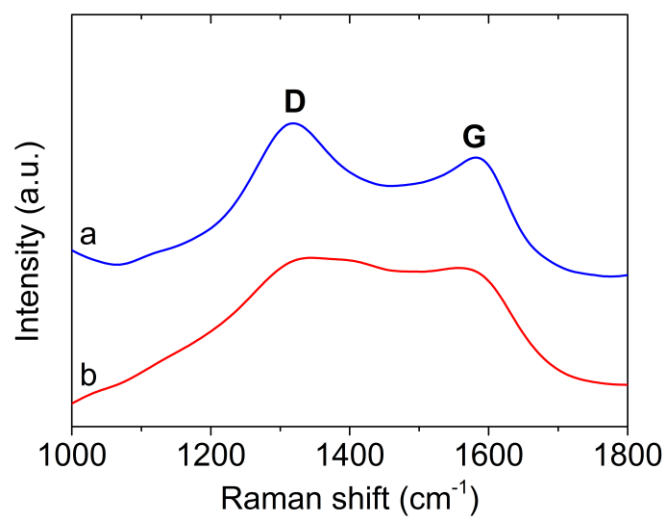


Figure S1: Experimental Raman spectra of Na<sub>2+x</sub>Ti<sub>4</sub>O<sub>9</sub>/C (a), and of amorphous carbon (b).

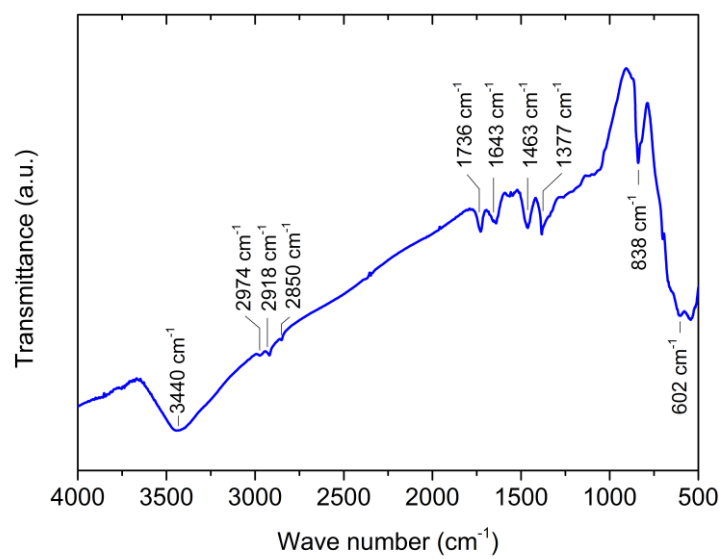


Figure S2: FTIR spectrum of Na<sub>2+x</sub>Ti<sub>4</sub>O<sub>9</sub>/C.

Table S1: Assignment of the peaks in Figure S2.

| Wave number (cm <sup>-1</sup> ) | Vibration assignment                    |
|---------------------------------|---|
| 3440                            | $\nu_s(\text{O-H}) \text{ H}_2\text{O}$ |
| 2974                            | $\nu_s(\text{C-H})$                     |
| 2918                            | $\nu_s(\text{C-H})$                     |
| 2855                            | $\nu_s(\text{C-H})$                     |
| 1736                            | $\nu_s(\text{C=O})$                     |
| 1643                            | $\nu_s(\text{C=O})$ amide               |
| 1463                            | $\nu_s(\text{C-N})$ amide               |
| 1377                            | $\nu_b(\text{C-O})$                     |
| 838                             | $\nu_b(\text{C-H})$                     |
| 602                             | $\nu_s(\text{Ti-O})$                    |

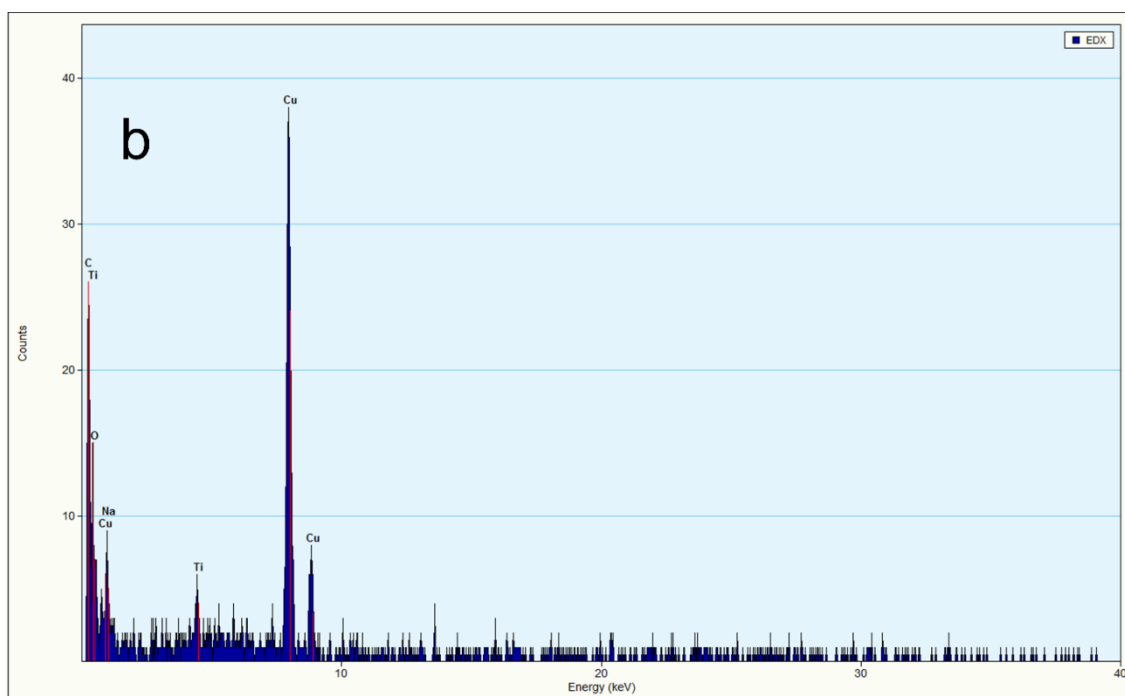
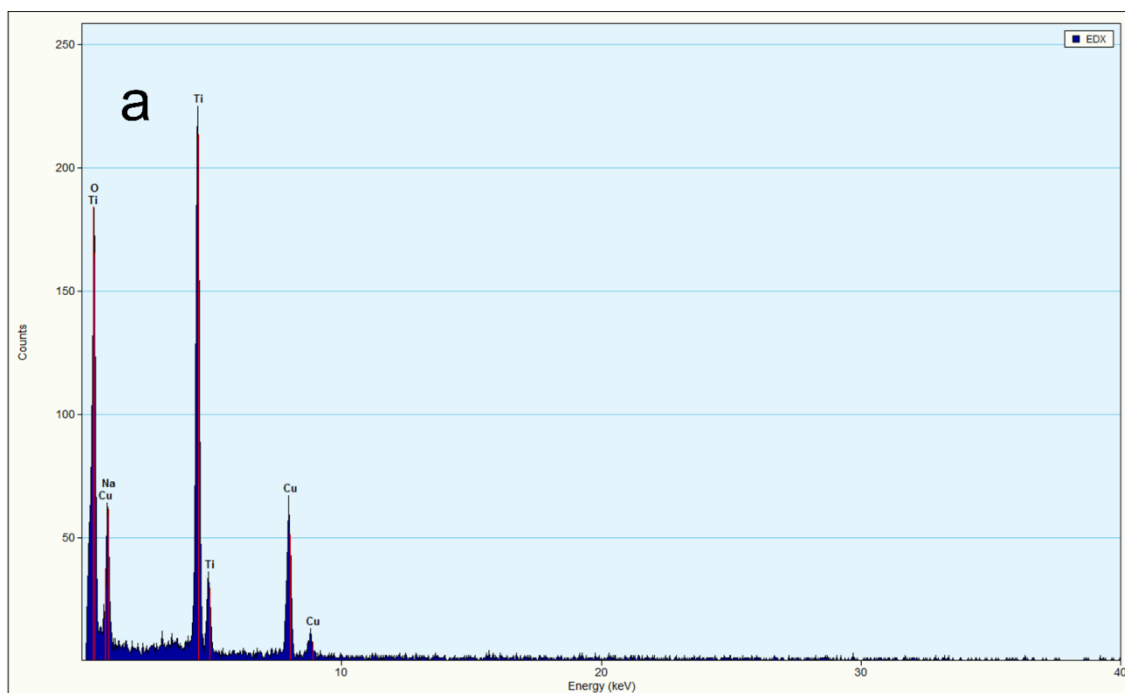


Figure S3: EDX of (a) a rod-shaped particle and (b) irregular nanoparticles found in  $\text{Na}_{2+x}\text{Ti}_4\text{O}_9/\text{C}$ . The Cu signal originates from the specimen support.

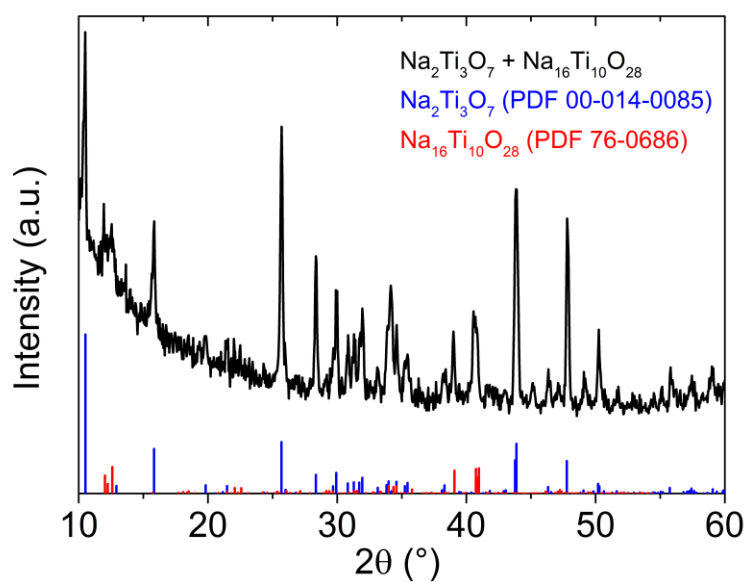


Figure S4: X-ray powder diffraction pattern of the  $\text{Na}^+ \text{-Ti}^{4+}$  precursor thermally processed for 2 hours under air, and references.

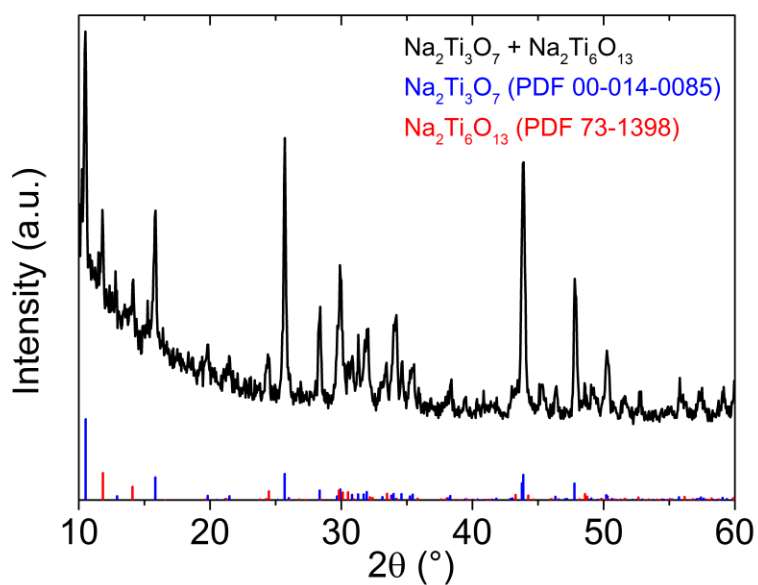


Figure S5: X-ray powder diffraction pattern of  $\text{Na}_{2+x}\text{Ti}_4\text{O}_9/\text{C}$  after thermal processing in air for 2 hours, and references.

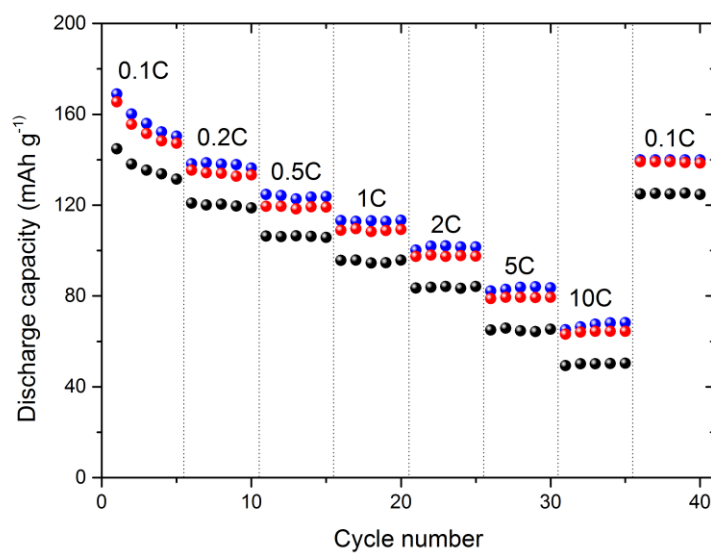


Figure S6: Individual discharge capacity measurements of  $\text{Na}_{2+x}\text{Ti}_4\text{O}_9$  at rates from 0.1C to 10C.

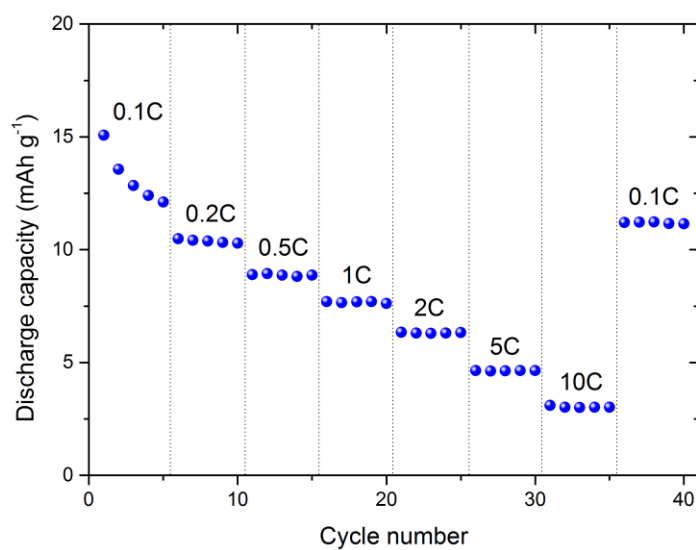


Figure S7: Discharge capacity of carbon black at rates from 0.1C to 10C. Whereas the slurries for these batteries contained only the additive carbon black and PVDF, the C-rates indicated in the graph are calculated according to a loading of  $2.55 \text{ mg cm}^{-2} \text{ Na}_{2+x}\text{Ti}_4\text{O}_9/\text{C}$ . This makes it possible to determine the capacity contribution of carbon black at every C-rate.

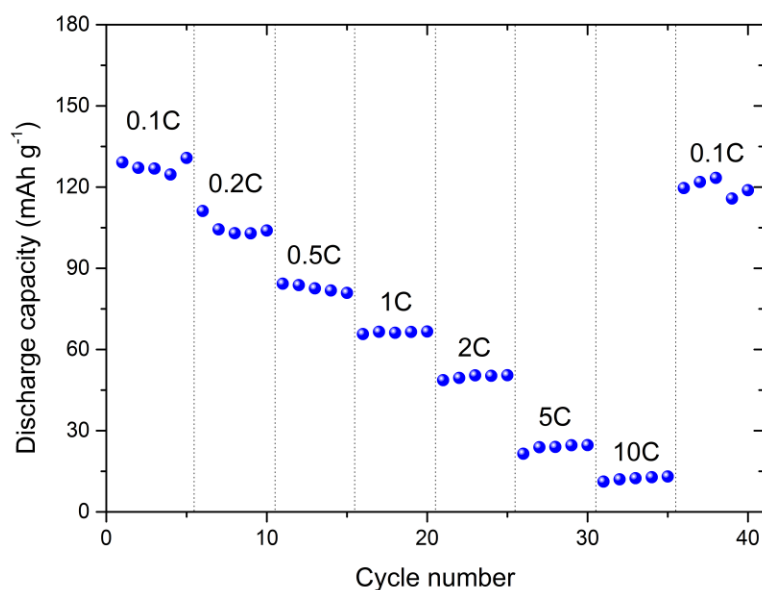


Figure S8: Discharge capacity of amorphous carbon at rates from 0.1C to 10C.

**Density functional theory calculations** Density functional theory (DFT) calculations on both  $\text{Na}_3\text{Ti}_4\text{O}_9$  (64 atoms per unit cell) and  $\text{Na}_2\text{Ti}_4\text{O}_9$  (60 atoms per unit cell), where the latter was characterized by introducing one Na vacancy per formula unit, were performed making use of the generalized gradient approximation (GGA), specifically of the Perdew-Burke-Ernzerhof (PBE) functional, implemented within a plane wave basis in the ABINIT code.<sup>47</sup> Electron-ion interactions were treated using norm-conserving pseudopotentials,<sup>48</sup> taking into account Na  $2s^2 2p^6 3s^1$ , Ti  $3s^2 3p^6 3d^2 4s^2$  and O  $2s^2 2p^4$  as valence electrons. The energy cutoff for the plane-wave basis was set to 40 Ha, to achieve convergence of the total energy below 1 meV per atom. The k-point grid used for integration over the Brillouin zone measured  $2 \times 15 \times 4$ . The notational convention established by Setyawan and Curtarolo (2010)<sup>49</sup> was used to denote the high-symmetry k-points. The crystal structure relaxation was performed using a conjugate-gradient algorithm ensuring that forces on each atom were below  $1 \text{ meV } \text{\AA}^{-1}$ .

The stoichiometric composition of  $\text{Na}_{2+x}\text{Ti}_4\text{O}_9$  is  $\text{Na}_3\text{Ti}_4\text{O}_9$ , with one titanium ion in the 3+ oxidation state. In the calculated band structure of  $\text{Na}_3\text{Ti}_4\text{O}_9$  (Figure S8a), the band at the Fermi level suggests a metal-like

behavior. Because of the gap between this band and the underlying band,  $\text{Na}_3\text{Ti}_4\text{O}_9$  is energetically unfavorable. By randomly creating  $\text{Na}^+$  vacancies, and thereby obtaining  $\text{Na}_2\text{Ti}_4\text{O}_9$ , the Fermi level is lowered and the material becomes an indirect semiconductor with a valence band maximum between  $\Gamma$  and Y and a conduction band minimum between X and  $X_1$  (Figure S8b). The band gap is found to amount to 2.2 eV. Due to the use of the semi-local PBE functional in the calculation, this gap may be underestimated.<sup>50</sup> Nevertheless, it is clear that this compound is not conductive.

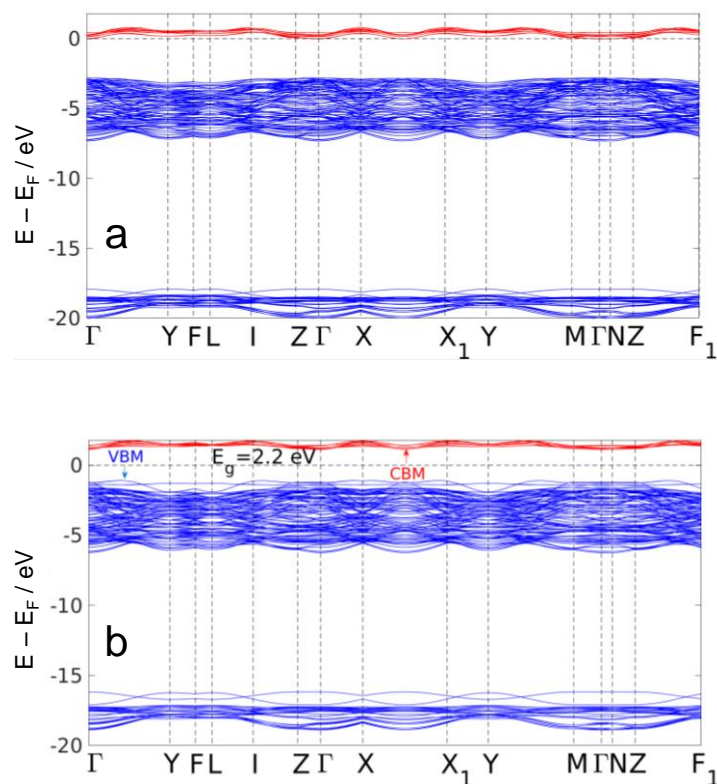


Figure S9: The calculated band structure along points of high symmetry in the irreducible Brillouin zone of stoichiometric  $\text{Na}_3\text{Ti}_4\text{O}_9$  (a) and  $\text{Na}_2\text{Ti}_4\text{O}_9$  (b). Valence bands are depicted as blue and conduction bands as red.



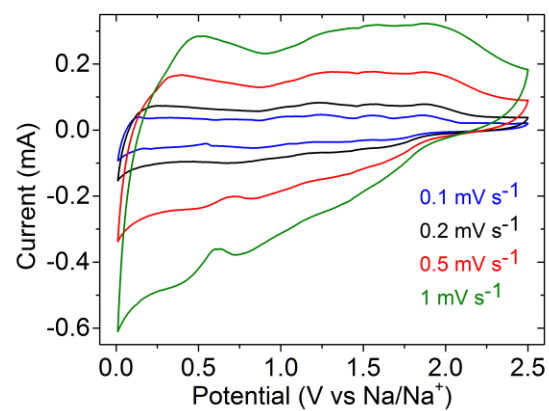


Figure S10: CV of a  $\text{Na}_{2+x}\text{Ti}_4\text{O}_9/\text{C} - \text{Na}$  cell at  $0.1 \text{ mV s}^{-1}$ ,  $0.2 \text{ mV s}^{-1}$ ,  $0.5 \text{ mV s}^{-1}$  and  $1 \text{ mV s}^{-1}$ .

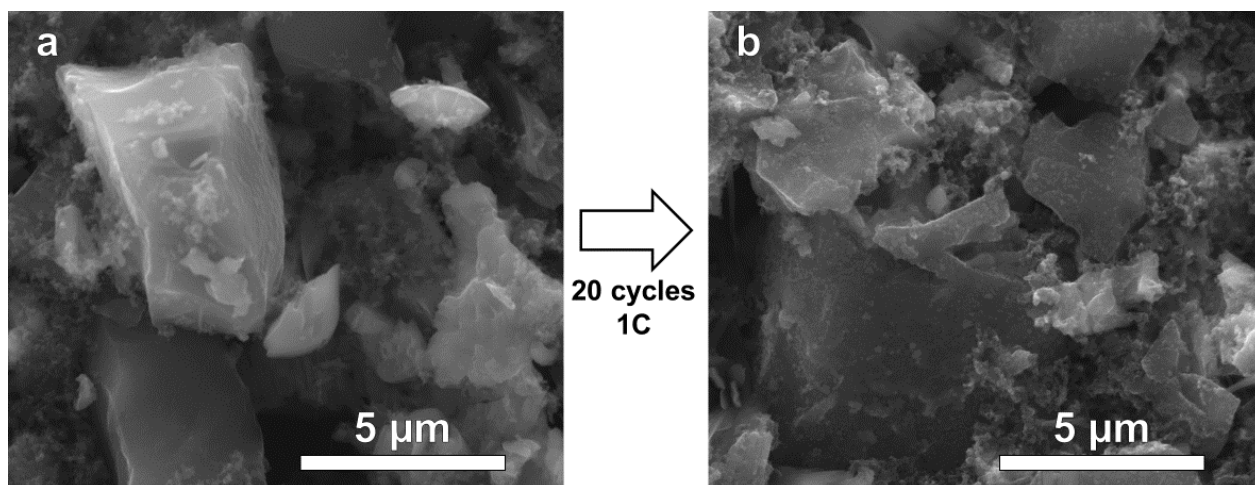


Figure S11: SEM images of  $\text{Na}_{2+x}\text{Ti}_4\text{O}_9/\text{C}$  electrodes: pristine (a) and after 20 charge/discharge cycles at 1C (b).

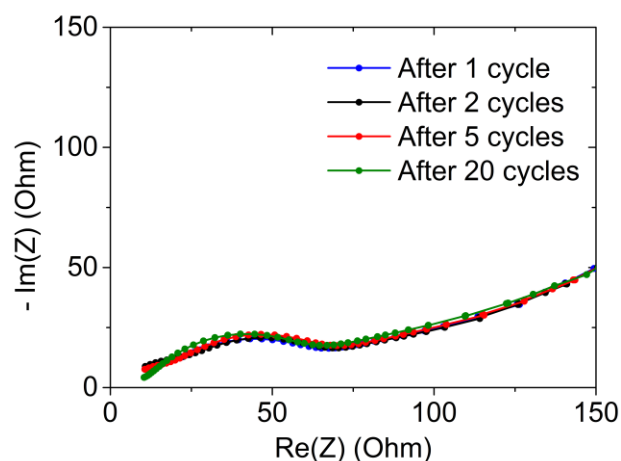


Figure S12: Nyquist plot representation of impedance spectra of  $\text{Na}_{2+x}\text{Ti}_4\text{O}_9/\text{C} - \text{Na}$  cells after 1, 2, 5 and 20 cycles at 1C (with an open circuit potential period of 2 hours after every cycle). The impedance spectra were recorded over a frequency range of 10 kHz to 10 mHz and subject to a potential amplitude of 10 mV.

## References

- (47) Gonze, X.; Amadon, B.; Anglade, P. M.; Beuken, J. M.; Bottin, F.; Boulanger, P.; Bruneval, F.; Caliste, D.; Caracas, R.; Côté, M.; et al. ABINIT: First-Principles Approach to Material and Nanosystem Properties. *Comput. Phys. Commun.* **2009**, *180* (12), 2582–2615.
- (48) Hamann, D. R. Optimized Norm-Conserving Vanderbilt Pseudopotentials. *Phys. Rev. B - Condens. Matter Mater. Phys.* **2013**, *88* (8), 1–21.
- (49) Setyawan, W.; Curtarolo, S. High-Throughput Electronic Band Structure Calculations: Challenges and Tools. *Comput. Mater. Sci.* **2010**, *54* (2), 299–312.
- (50) Aulbur, W. G.; Jönsson, L.; Wilkins, J. W. Quasiparticle Calculations in Solids. *Solid State Phys.* **1999**, *54*, 1–218.